Thermodynamic database development based on ab initio calculations: challenges and opportunities

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Outline

- Thermodynamic database content
- Database development
 - Based on experiments
 - Based on ab initio
- Challenges
- Opportunities



Thermodynamic database content

- ΔH²⁹⁸
- S²⁹⁸
- C_p(T) **G(T)**
- µ
- T_C

Energy: Joules Pressure: a	m Al	– 🗆 X
<u>File E</u> dit <u>U</u> nits <u>V</u> iew <u>T</u> o	ols View <u>D</u> ata <u>H</u> elp	
Formula		
⊢r 🗅 FTliteBASE	S1 properties • Heat of form. + Entropy • O Heat + Temperature	of transf.
⊡- <mark>∠ S1</mark> -Cp 700	Form. of S1	K)) 35571
-Cp 3900 ⊡-∠ S2	Phase Name Reference no. De fcc_A1 501 2.6	nsity g/cc 9822
⊡ <mark> </mark> S3	Extended properties (optional)	
⊡⊸L∠ S4	Birch-Murnaghan	
⊡ ⊑ 000 ⊡ ⊑ S6	Therm. expans. (/K) Compressibility (/bar) Bulk r	nod. deriv.
⊡	7.6228E-8 T T	
⊡		(1-10)//(1710)
	0.00857216 /I ² T ³	
r 🛱 FTmiscBASE	Magnetic	
r 🗅 FToxidBASE	Moment (D) Temperature P fac © Curie K © 0.2 © Neel K	tor 28 40
🚽 r 🗅 FTsaltBASE		
FactSage 7.1 C:\F/	CTSAGE71\FACTDATA\FTLITE60BASE.CDB (v7.10) 1410 comp	ounds read-only



Thermodynamic database content

•	ΛH ²⁹⁸	Energy: Joules Pressure: atm Al	- [×						
_	O 208	<u>File Edit Units View Tools ViewData H</u> elp									
•	5290										
•	C _p (T) G(T)	Al Form. of S1 → Al → Al → Cp 700	S1 properties Heat of form. + Entropy C Heat + Temperature of transf. Form. of S1 <u>AH 298 (Joules)</u> S298 (J/(mol K))								
•	μ	Cp 933 Cp 3900 Phase Name Reference no.	35571 ensity g/cc								
•	T _C	B S2 Extended properties (optional)	39822]							
	0	Birch-Murnaghan									
•	ρ	Therm. expans. (/K) Compressibility (/bar) Bulk r	mod. deriv.								
	$\alpha(\mathbf{T})$	⊕∠ S7									
•	u(1) G(Tp)	⊕ <u>L</u> S8 7.6228E-8 T T	(T-	T0)In(T/T0)						
•	К(Т)	$\square \square $									
•	B'(T)	r C FTOxCNBASE r C FToxidBASE r C FTpulpBASE r C FTsaltBASE ↓ Magnetic Moment (D) Temperature P fac C Ourie K C O. C Neel K C O.	Magnetic Moment (D) Curie Neel K C 0.40								
		FactSage 7.1 C:\FACTSAGE71\FACTDATA\FTLITE60BASE.CDB (v7.10) 1410 com	pounds read	-only							



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- C_p(T) **G(T)**
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- α(T) (T) G(T,p)
- K(T)
- B'(T)
- L(T) **G(T,p,X**_i)





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Database development based on experiment

- Ideal: The following properties are measured:
 - binary phase diagram
 - enthalpies of (trans-)formation of the compounds
 - enthalpies of mixing and constituent activities of solutions as f(T)
 - heat capacities of compounds and solutions
 - crystal structures, especially for complex solutions
 - density as f(T)
 - elastic properties as f(T)
- Real: Few of the properties are measured.



Database development based on experiment

- Experimental error bars can be evaluated to fit the
- Ideal: thermodynamic properties to all measurements
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- 2nd step: Consistency checks of binary and ternary systems!
- elas Thermodynamic properties are fitted to all available measurements



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→ materialsproject.org, oqmd.org, aflowlib.org, nomad-coe.eu



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Challenge: $\Delta H_{f,experiment} \leftrightarrow \Delta H_{f,ab initio}$

Publications including "lattice stability"





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Mean average error is ~13 kJ/mol atoms

Kirklin et al. (2015), The Open Quantum Materials Database (OQMD): assessing the accuracy of DFT formation energies

(1 eV/atom ≈ 100 kJ/mol atoms)



Challenge: $\Delta H_{f,experiment} \leftrightarrow \Delta H_{f,ab initio}$



Linear correction for elements:



Mean average error is ~9 kJ/mol atoms

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(1 eV/atom \approx 100 kJ/mol atoms)



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Linear correction for elements:





Challenge: "High" temperatures

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Formula			
r Cp 3900 r C S1 r S2 r C S1 r S2 r S1 r S2 r S1 r S2 r S1 r S2 r S2 r S2 r S3 r S2 r S3 r S2 r S3 r S2 r S3 r S5 r S3 r S5 r S3 r S5 r S3 r S5 r S3 r S5 r S5	Cp expression for S1 AH 298: 0 J/mol S 298: 28.2999967235571 J/(mol K) from 298.150 K to 700.000 K H Cp(T) = 24.3671976 T^ 0.00 0.003769324 T^ 1.00 G Edit -148184 T^ -2.00 5.265984E-6 T^ 2.00 T^ 1 T^		
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Challenge: "High" temperatures

Ab initio calculations are mostly done at 0K.

How to extrapolate to room temperature and above?

- → Einstein model
- \rightarrow Debye model
 - \rightarrow X C_p = const X
- → Other models for $T > \theta_D$ are computationally expensive!





Challenge: Liquid and gas phases

• There is currently no high-throughput solution in ab initio calculations...



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Opportunities: Unstable allotropes

Different allotropes are often modelled with the same $C_p(T)$ -function.





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C_{p,BCC-Al} should be slightly higher!

There is a disagreement of S_{FCC-AI} between SGTE and ab initio calculations at 298 K.

Ab initio calculations can help improving properties that are hard to measure!



-SGTE_FCC —SGTE_BCC



Opportunities: Enthalpies at low T

• It is impossible to measure enthalpies at low temperatures due to kinetic constraints.



Barzilai, Acta Materialia 120 (2016) 255.



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"The predicted critical temperature is very close to the lower limit of the temperature range of currently known experimental phase diagrams."

Barzilai, Acta Materialia 120 (2016) 255.



Opportunities: Navigating Chemical Compound Space

Today: SGTE Solution database contains ~1500 phases. That allows description of

577 binary systems,

141 ternary systems,

15 higher order systems



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Н	141 ternary systems,															Не	
Li	Ве	^{Be} 15 higher order systems									В	С	Ν	0	F	Ne	
Na	Mg	i o nigrici oraci systems								AI	Si	Р	S	CI	Ar		
κ	Ca	Sc	Ti	V	Cr	Mn	Fe	Со	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Υ	Zr	Nb	Мо	Тс	Ru	Rh	Pd	Ag	Cd	In	Sc	Sb	Те	Т	Хе
Cs	Ва	La	Hf	Та	W	Re	Os	Ir	Pt	Au	Hg	TI	Pb	Bi	Ро	At	Rn
Fr	Ra	Ac	Rf	Db	Sg	Bh	Hs	Mt	Ds								

~70 non-radioactive, non-noble gas elements \rightarrow 70*69 = 4830 binary systems \rightarrow 70*69*68 = 328440 ternary systems



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Li	Ве	15 higher order systems	В	С	Ν
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materialsproject.org, oqmd.org, aflowlib.org, nomad-coe.eu $69'623 + 471'857 + 1'672'887 \rightarrow 17'150'125$ compounds (for most of them only ΔH_f^{0K})

> ~70 non-radioctive, non-noble gas elements \rightarrow 70*69 = 4830 binary systems \rightarrow 70*69*68 = 328440 ternary systems



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Thank you very much for your attention!

