

# The GTT Aluminium database

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# Introduction:

GTT-Technologies

## Why an additional Al-database?

- starting from existing database (former project, exp. results)
- several metastable phases in Al-Mg-Si system considered by Povoden-Karadeniz et al.<sup>1</sup> (great importance for production of light-weight alloys)
- more emphasis on experimental results ( $c_p$ -functions)



<sup>1</sup>E. Povoden-Karadeniz, P. Lang, P. Warczok, A. Falahati, W. Jun, E. Kozeschnik, Calphad 43 (2013) 94

# Database:

## Contents:

elements 8:

Al, Cu, Fe, Mg, Mn, Ni, Pb, Si

binary systems (28/28):

Al-Cu, Al-Fe, Al-Mg, Al-Mn, Al-Ni,  
Al-Pb, Al-Si, Cu-Fe, Cu-Mg, Cu-Mn,  
Cu-Ni, Cu-Pb, Cu-Si, Fe-Mg, Fe-Mn,  
Fe-Ni, Fe-Pb, Fe-Si, Mg-Mn, Mg-Ni,  
Mg-Pb, Mg-Si, Mn-Ni, Mn-Pb,  
Mn-Si, Ni-Pb, Ni-Si, Pb-Si,  
(Al-Zr) optimised



# Database:

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**ternary systems (10/56):**

**Al-Cu-Mg, Al-Cu-Si, Al-Fe-Mg,  
Al-Fe-Ni, Al-Mg-Si, Cu-Fe-Ni,  
Al-Fe-Mn, Al-Fe-Si, Al-Mg-Mn,  
Al-Mn-Si**

**ternary systems containing Al (9/21) ⇒ still not complete**

**solution phases:** 65

**stoichiometr. condensed phases:** 52

**reference states for elements (s, l)**



# Database:

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metastable Al,Mg,Si phases contained in the database<sup>1</sup>:

CL_MGSI	MgSi-co-clusters	(Mg,Si) <sub>1</sub>
GP_MAT	Guinier-Preston-zones (GP) “Matsuda-phase” ordered FCC phase	(Al,Mg,Si) <sub>1</sub> (Al,Mg,Si) <sub>1</sub> (Al,Mg,Si) <sub>1</sub> (Al,Mg,Si) <sub>1</sub>
Al2Mg5Si4	Al-β''	(Al) <sub>2</sub> (Mg) <sub>5</sub> (Si) <sub>4</sub>
Al4Mg8Si7	B' <sub>H</sub>	(Al) <sub>4</sub> (Mg) <sub>8</sub> (Si) <sub>7</sub>
Al3Mg9Si7	B' <sub>L</sub>	(Al) <sub>3</sub> (Mg) <sub>9</sub> (Si) <sub>7</sub>
Mg9Si5	β'	(Mg) <sub>9</sub> (Si) <sub>5</sub>
Mg5Si6	β''	(Mg) <sub>5</sub> (Si) <sub>6</sub>
u1_phase	u1	(Al) <sub>2</sub> (Mg) <sub>1</sub> (Si) <sub>2</sub>
u2_phase	u2	(Al) <sub>1</sub> (Mg) <sub>1</sub> (Si) <sub>1</sub>

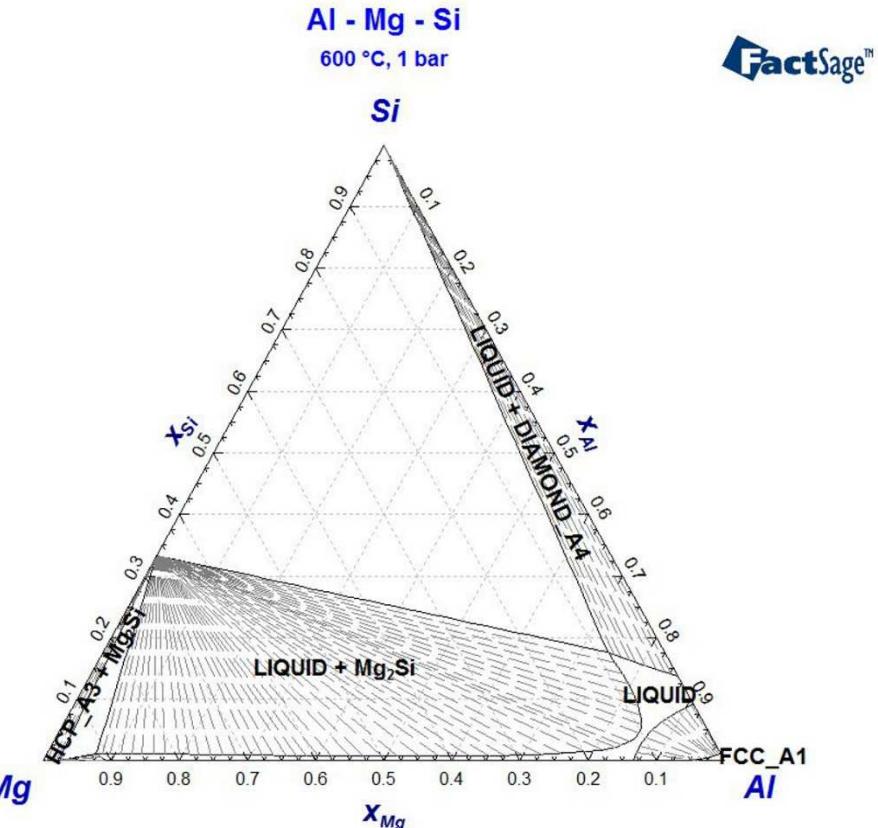
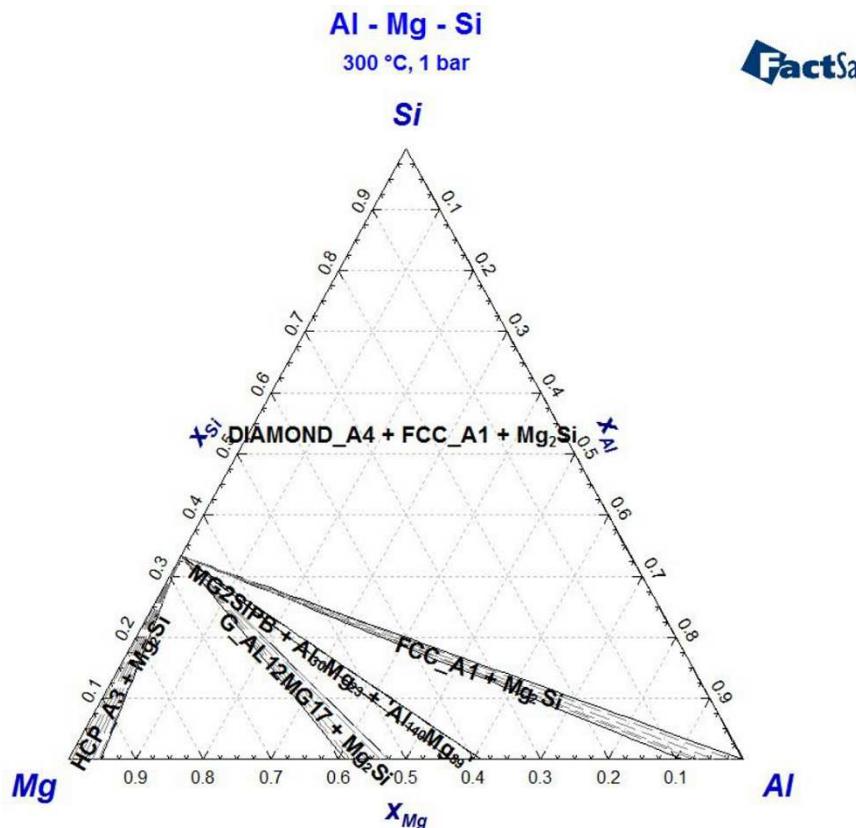


<sup>1</sup>E. Povoden-Karadeniz, P. Lang, P. Warczok, A. Falahati, W. Jun, E. Kozeschnik,  
Calphad 43 (2013) 94

# The ternary Al-Mg-Si system:

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isothermal cross sections (300°C, 600°C)



dominating phase: Mg<sub>2</sub>Si



# The ternary system Al-Fe-Ni

**ternary system Al-Fe-Ni neither in SGTE nor FSstel**

**Thermodynamic assessment from literature:**



Available online at [www.sciencedirect.com](http://www.sciencedirect.com)



Acta Materialia 57 (2009) 5324–5341



[www.elsevier.com/locate/actamat](http://www.elsevier.com/locate/actamat)

Thermodynamic properties of the Al–Fe–Ni system acquired  
via a hybrid approach combining calorimetry,  
first-principles and CALPHAD

Lijun Zhang<sup>a</sup>, Jiong Wang<sup>a</sup>, Yong Du<sup>a,\*</sup>, Rongxiang Hu<sup>b</sup>, Philip Nash<sup>b</sup>,  
Xiao-Gang Lu<sup>c</sup>, Chao Jiang<sup>a</sup>

<sup>a</sup> State Key Laboratory of Powder Metallurgy, Central South University, Changsha, Hunan 410083, China

<sup>b</sup> Thermal Processing Technology Centre, Illinois Institute of Technology, 10 West, 32nd Street, Chicago, IL 60616, USA

<sup>c</sup> Thermo-Calc Software AB, Björnnäsvägen 21, 113 47 Stockholm, Sweden

Received 26 March 2009; received in revised form 24 May 2009; accepted 19 July 2009

Available online 21 August 2009

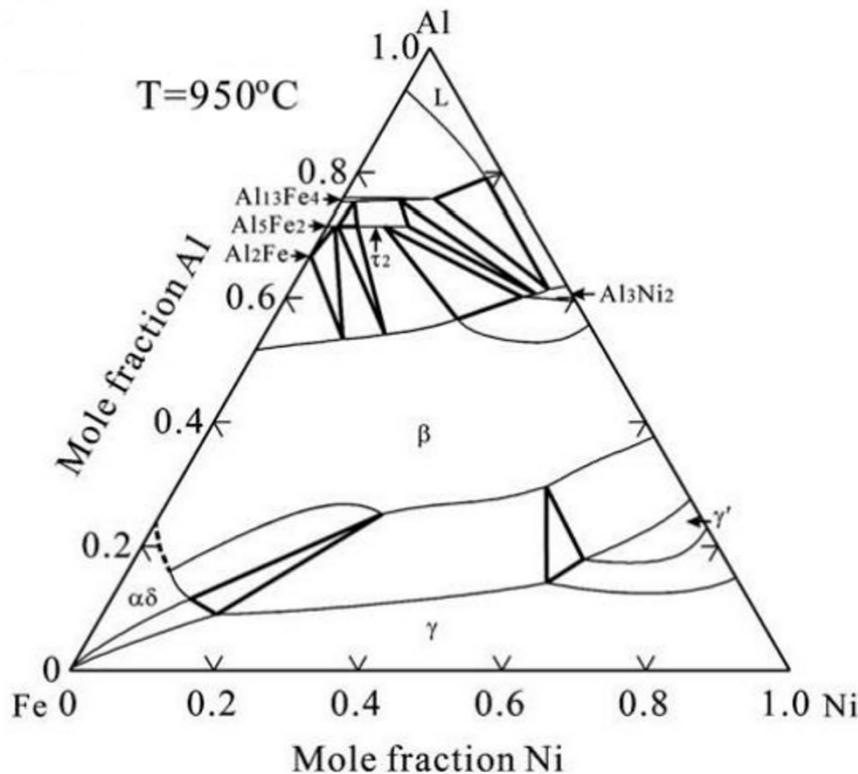
**incorporation into database after critical inspection!**



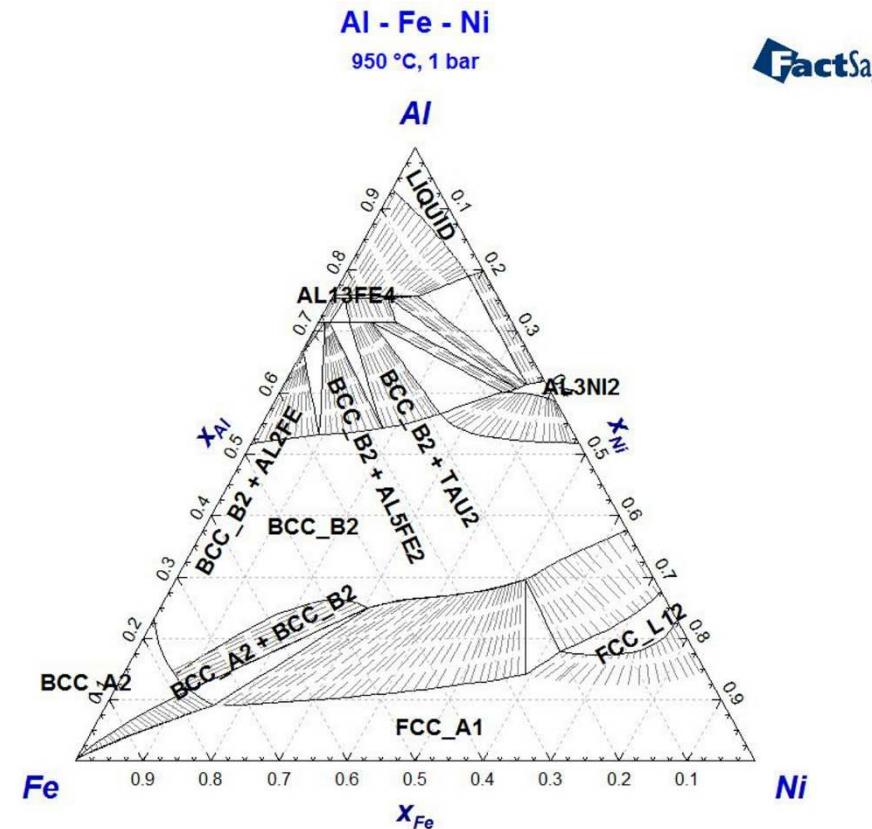
# The ternary system Al-Fe-Ni

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## comparison of isothermal sections:



Zhang et al. (2009)

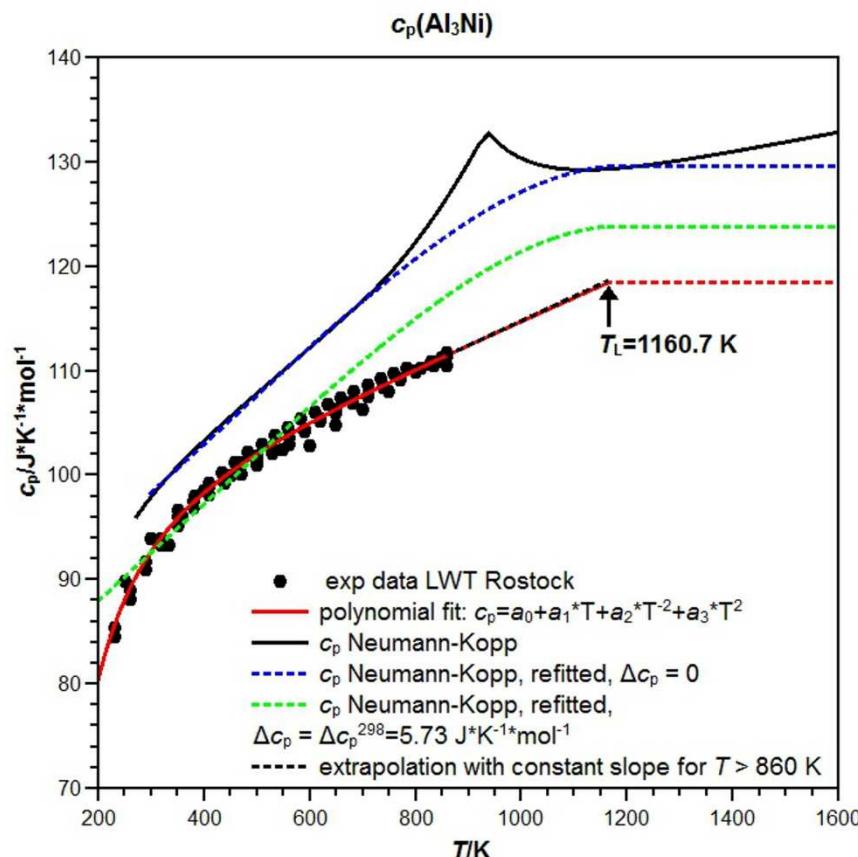


calculated from database

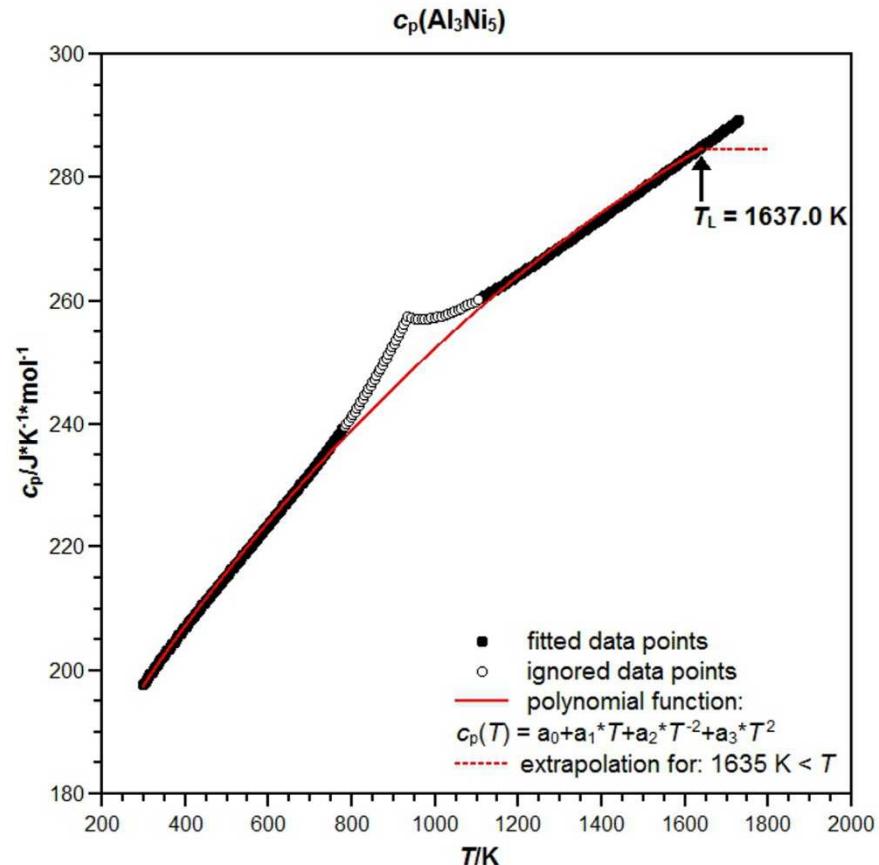
# Experimental data for $c_p$ :

## Al-Ni binary system:

$c_p(\text{Al}_3\text{Ni})$ :  
(LWT Rostock)

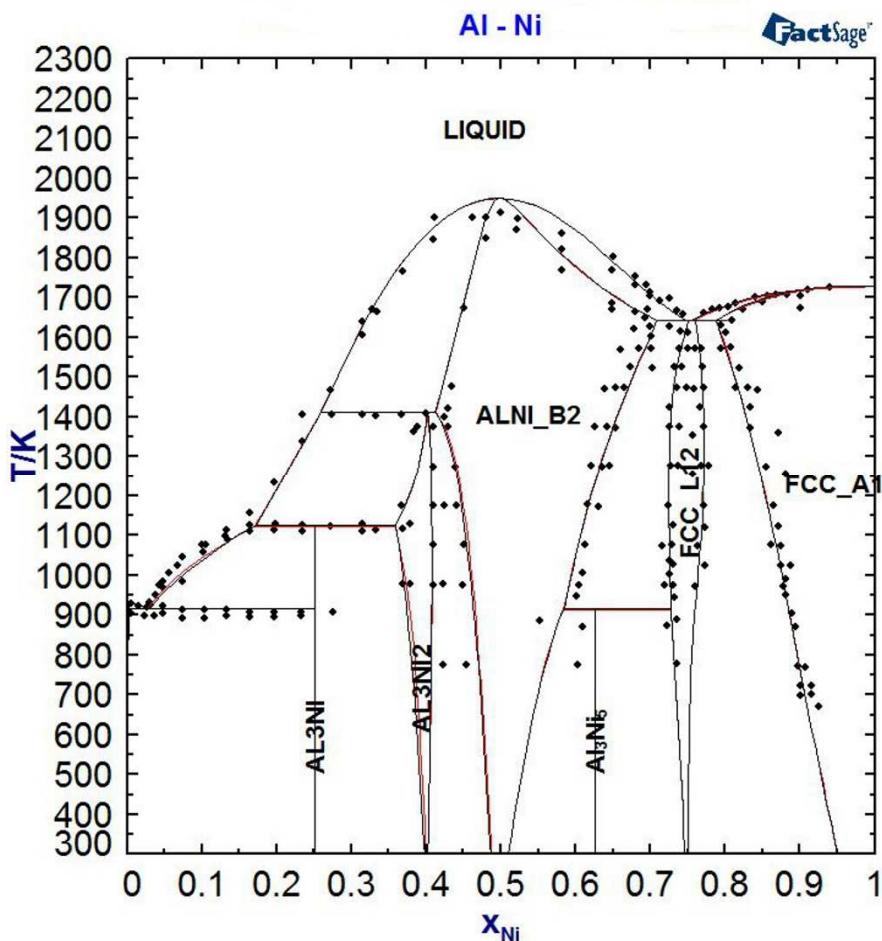


$c_p(\text{Al}_3\text{Ni}_5)$ :  
refitted Neumann-Kopp



# Experimental data for $c_p$ :

reoptimization of binary Al-Ni:



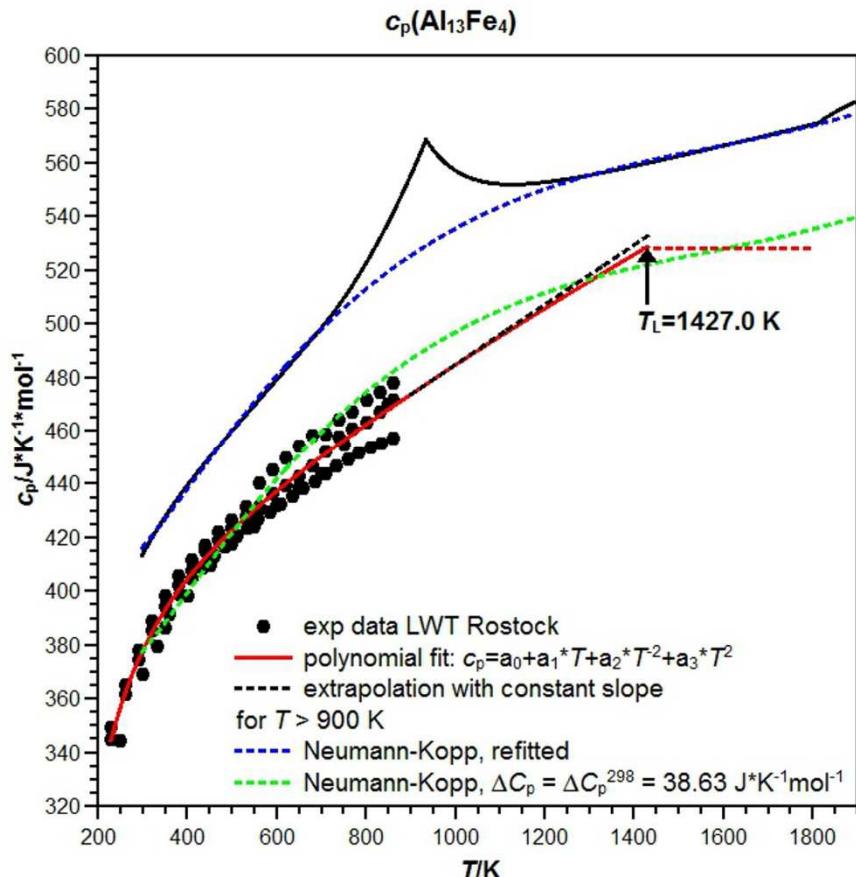
comparison between:

initial phase diagram  
(black),  
optimised phase diagram  
(red)  
experimental data



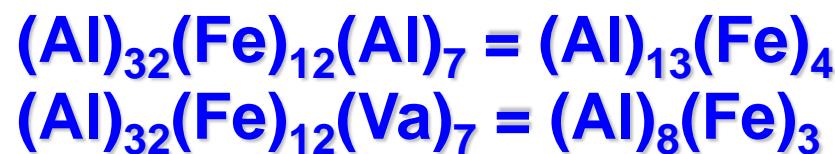
# Experimental data for $c_p$ :

## Al-Fe binary system:



**AL13FE4 is non-stoichiometric (solution phase), modelled  $(\text{Al})_{32}(\text{Fe})_{12}(\text{Al},\text{Va})_7$ ,**

**end-members:**



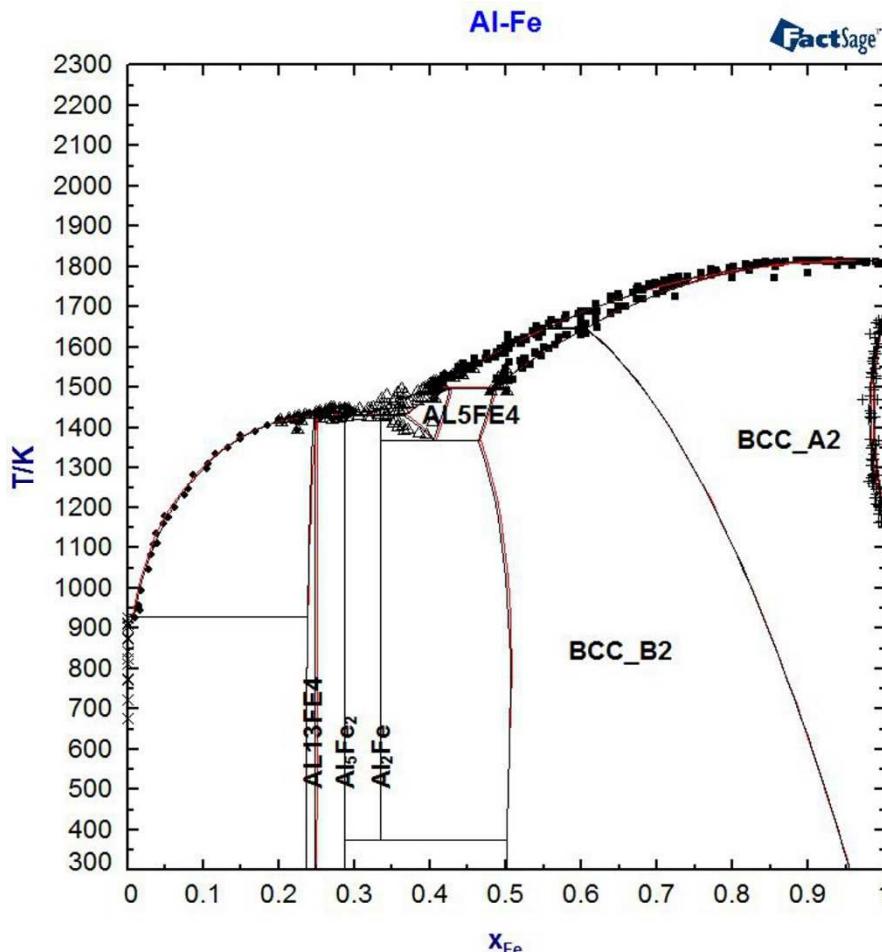
**Assumption (within experimental precision):**

$c_p(\text{Al}_8\text{Fe}_3)/11 = c_p(\text{Al}_{13}\text{Fe}_4)/17$   
per moles of atoms



# Experimental data for $c_p$ :

reoptimization of binary Al-Fe:



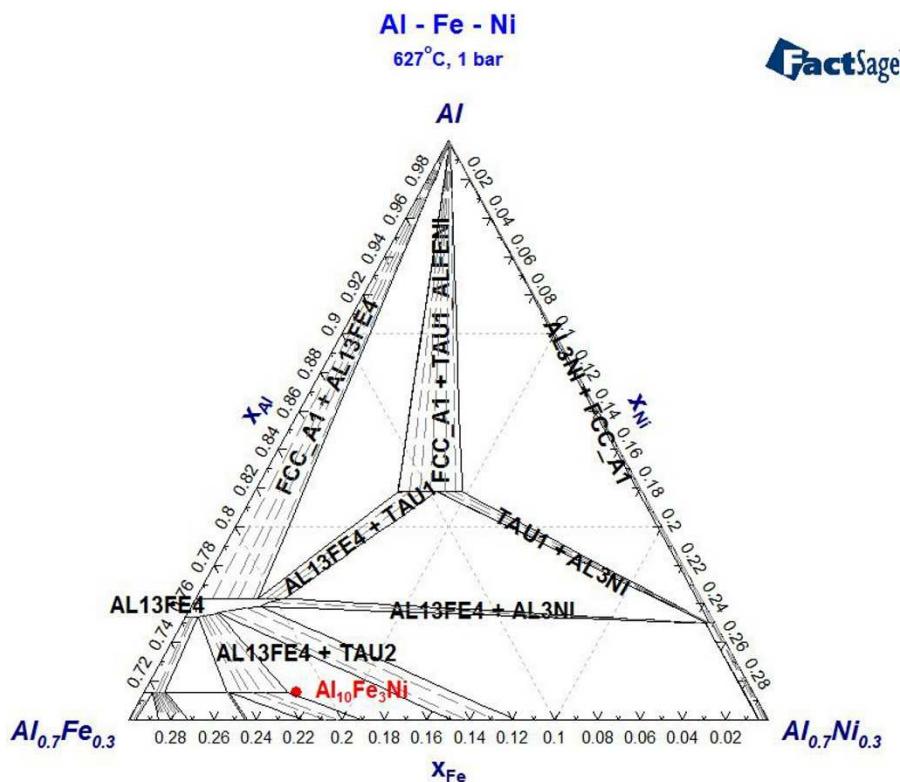
comparison between:

initial phase diagram  
(black),  
optimised phase diagram  
(red)  
experimental data



# Experimental data for $c_p$ :

$c_p$ -measurement for  $\text{Al}_{10}\text{Fe}_3\text{Ni}$  (TAU2) (LWT Rostock)

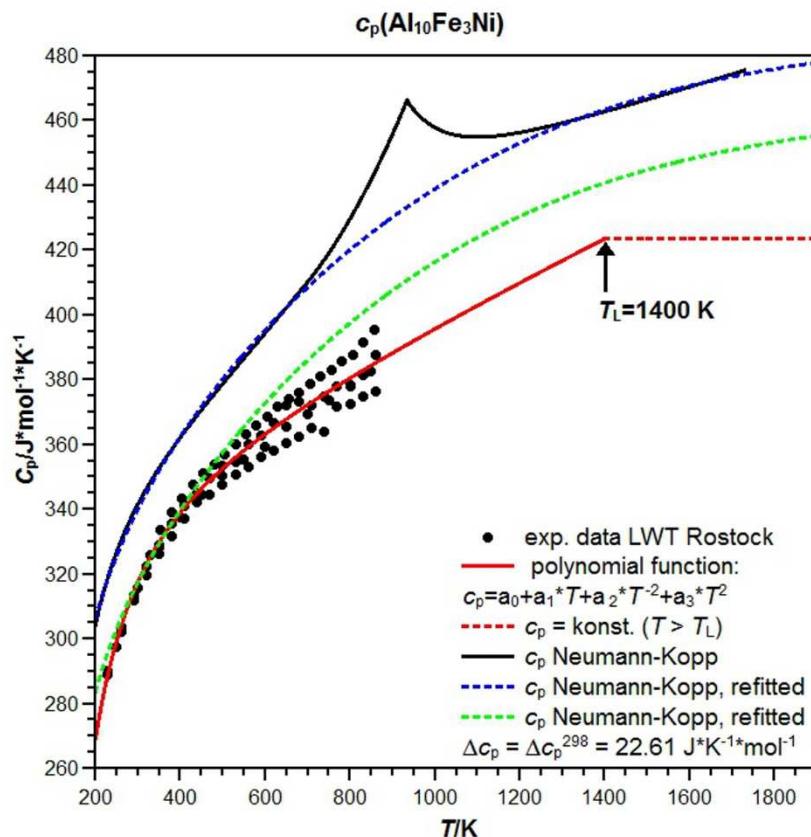


**TAU2:**  
model  $(\text{Al})_5(\text{Fe},\text{Ni})_2$   
end members:  
 $\text{Al}_5\text{Fe}_2 - \text{Al}_5\text{Ni}_2$



# Experimental data for $c_p$ :

TAU2:  $(\text{Al})_5(\text{Fe},\text{Ni})_2$   
 end members:  $\text{Al}_5\text{Fe}_2 - \text{Al}_5\text{Ni}_2$



same assumption as  
 for AL13FE4:

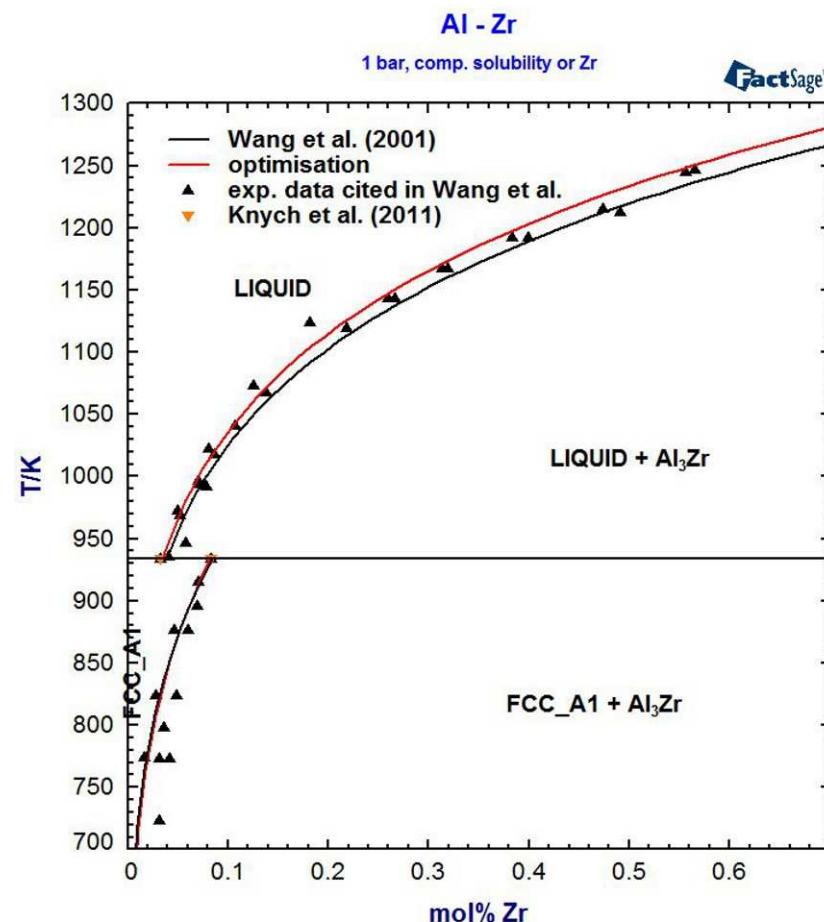
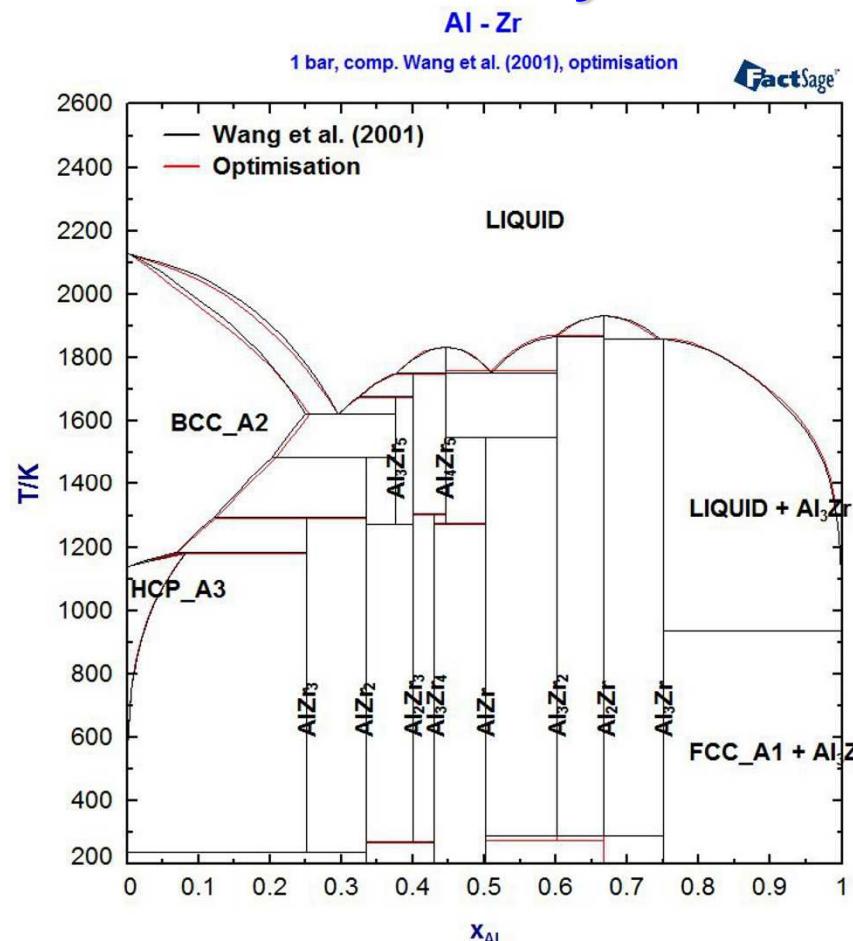
$c_p(\text{Al}_5\text{Fe}_2)/7 = c_p(\text{Al}_5\text{Ni}_2)/7 =$   
 $c_p(\text{Al}_{10}\text{Fe}_3\text{Ni})/14$   
 per mole of atoms

Reoptimization of ternary  
 Al-Fe-Ni system with  
 experimentally based  $c_p$ -  
 functions in progress



# Experimental data for $c_p$ :

**binary system Al-Zr optimized with exp.  $c_p(\text{Al}_3\text{Zr})^2$  and solubility of Zr in Al(fcc) and LIQUID<sup>3</sup>**



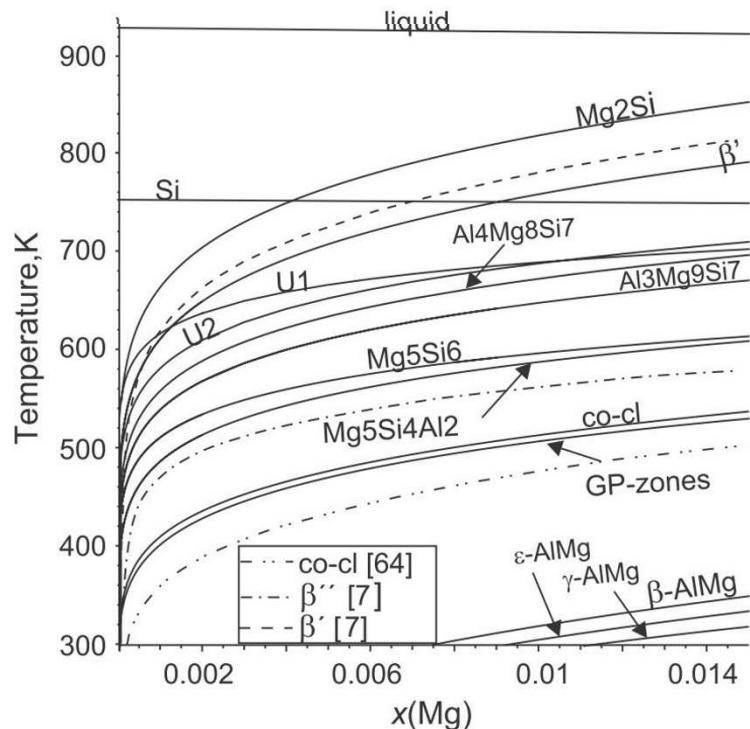
<sup>2</sup>T. Knych et al., Archives Met. Mater. 56(3) (2011) 685

<sup>3</sup>LWT Rostock

# Metastable phases in Al-Mg-Si :

**metastable equilibria in Al-Mg-Si system of great importance for precipitation hardening<sup>1</sup>:**

**supersaturated solid solution (rt) → clusters → co-clusters → GP-zones →  $\beta''$  →  $\beta'$ ,  $\beta'_H$ ,  $\beta'_L$ ,  $u_1$ ,  $u_2$**



**solfi of metastable phases compared to Mg<sub>2</sub>Si for an AA6016 alloy (0.76at% Si)<sup>2</sup>**

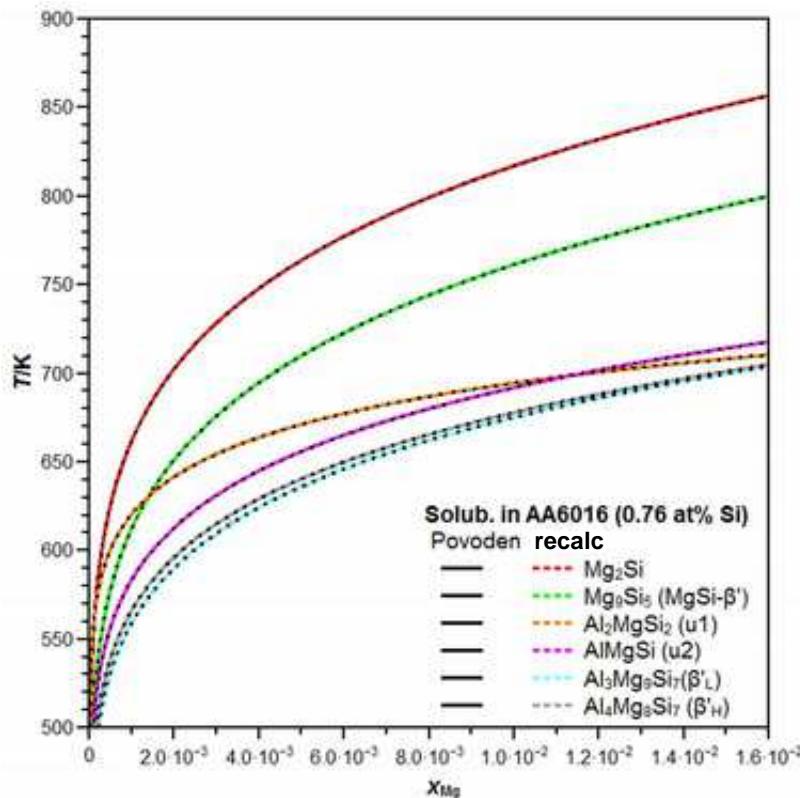


<sup>1</sup>E. Povoden-Karadeniz, P. Lang, P. Warczok, A. Falahati, W. Jun, E. Kozechnik, Calphad 43 (2013) 94

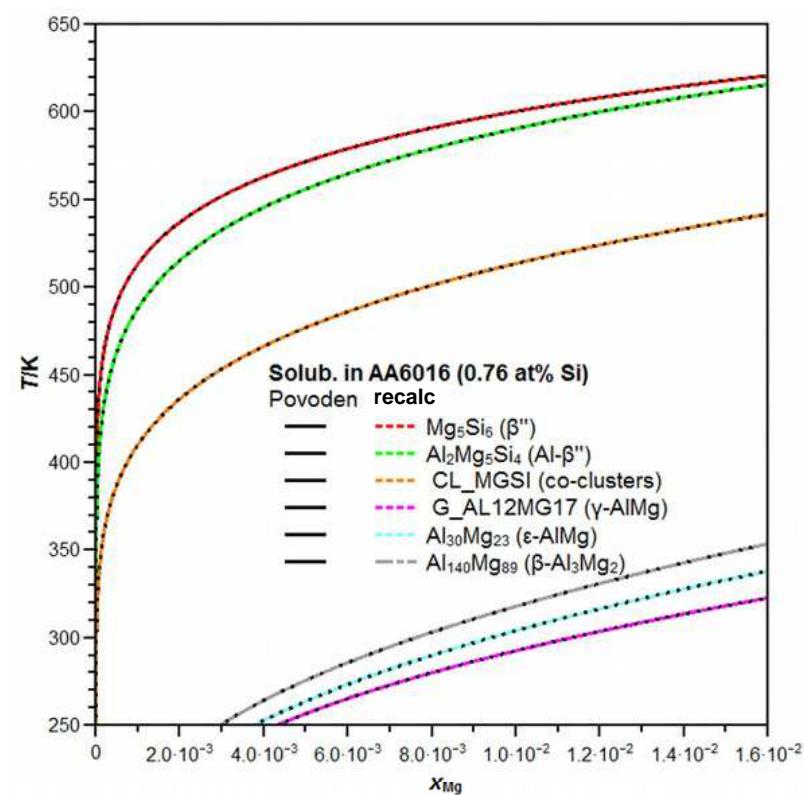
# Metastable phases in Al-Mg-Si :

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recalculation of solvi in Al(fcc),  
comp. to Povoden-Karadeniz et al.



stable phase:  $Mg_2Si$



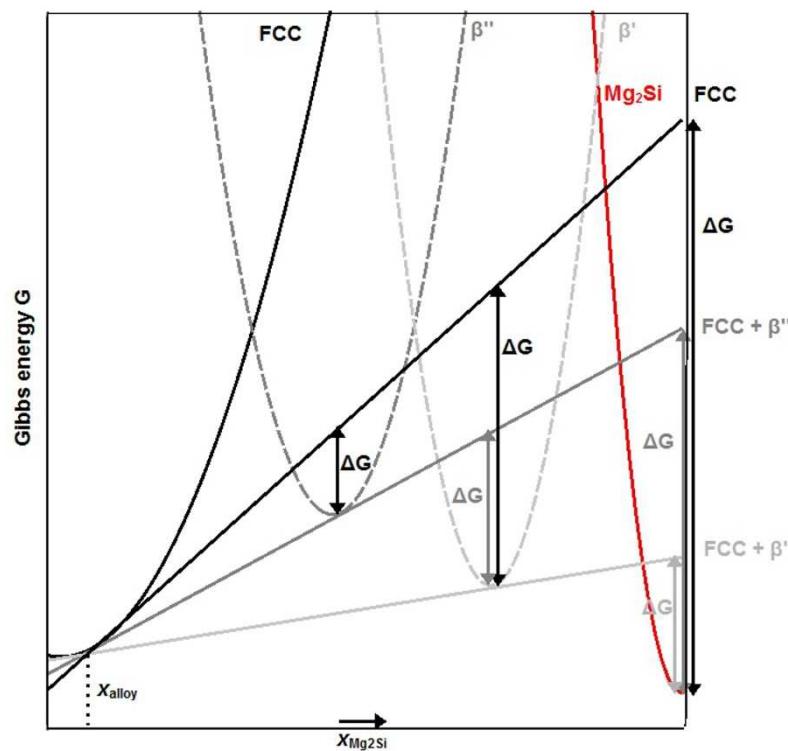
stable phases:  $\beta$ -,  $\gamma$ -,  $\epsilon$ -AlMg



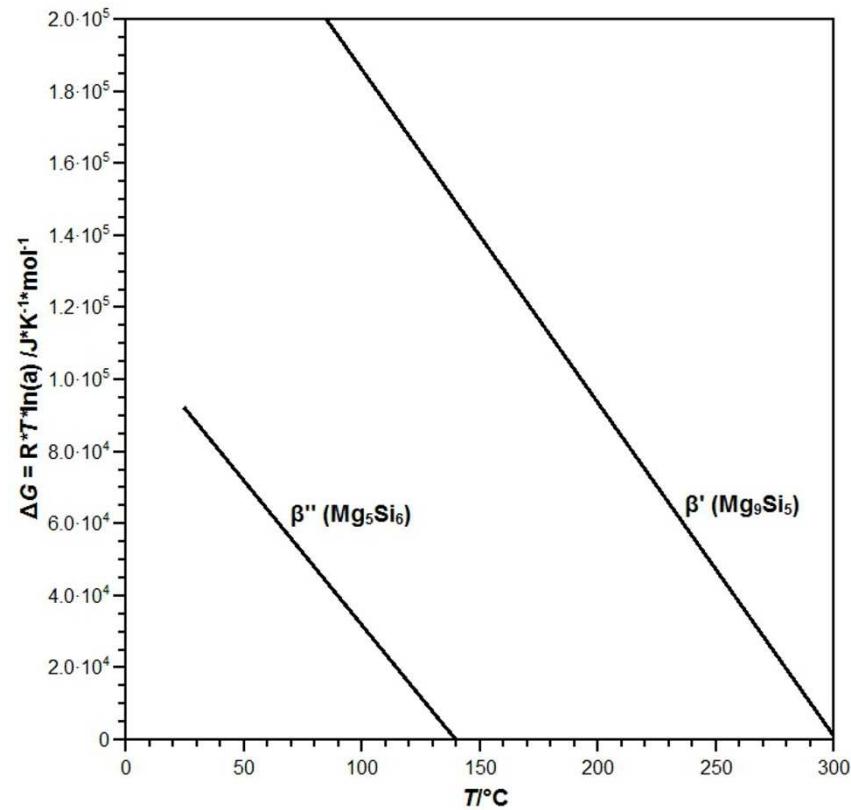
# Metastable phases in Al-Mg-Si :

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Gibbs energy diagram of selected metastable phases (sketch)



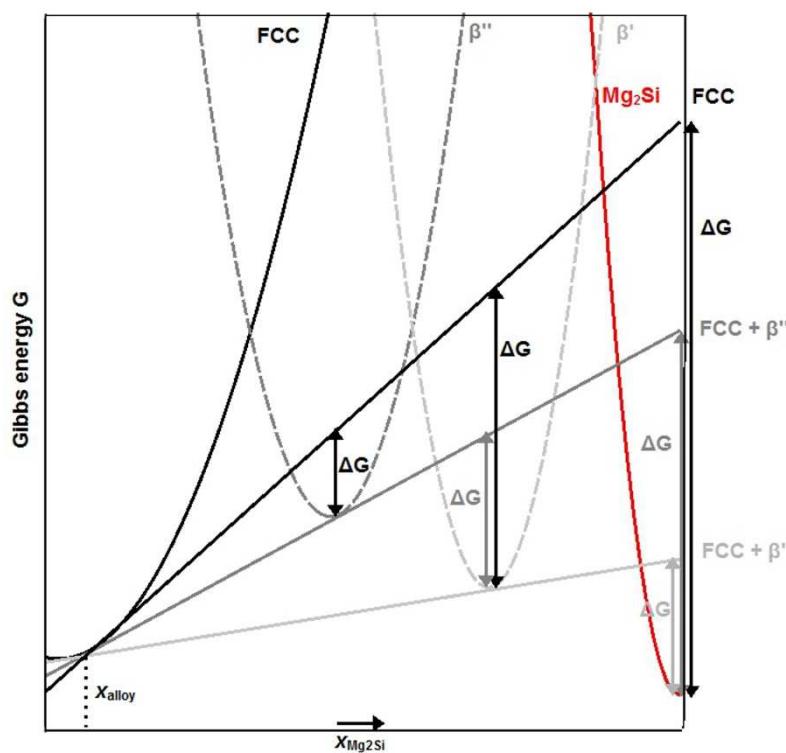
driving forces  
only FCC present



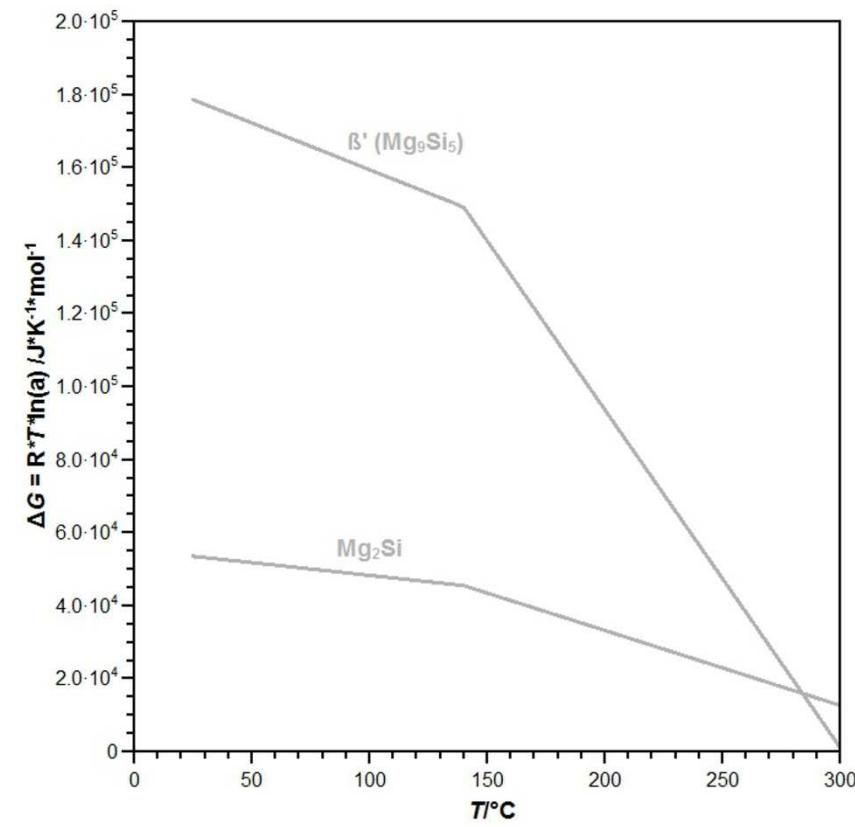
# Metastable phases in Al-Mg-Si :

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Gibbs energy diagram of selected metastable phases (sketch)



driving forces,  
FCC +  $\beta''$  present



# Summary:

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- **Actual contents of database:**  
**8 elements, 28 binary-, 10 ternary systems, 65 solution phases, 52 stoichiometr. compounds, metastable phases (Al-Mg-Si: 9)**
- **New thermodynamic assessment for Al-Fe-Ni (Zhang et al. 2009)**
- **Use of exp.  $c_p(T)$  for  $\text{Al}_3\text{Ni}$ , “ $\text{Al}_{13}\text{Fe}_4$ ”,  $\text{Al}_3\text{Zr}$ , “ $\text{Al}_{10}\text{Fe}_3\text{Ni}$ ” (TAU2)  $\Rightarrow$  reoptimization of Al-Ni, Al-Fe, Al-Zr binary systems, Al-Fe-Ni ternary system ongoing**
- **Calculations using the database (examples), metastable phases in the Al-Mg-Si ternary system**



**Thanks for your attention !**

