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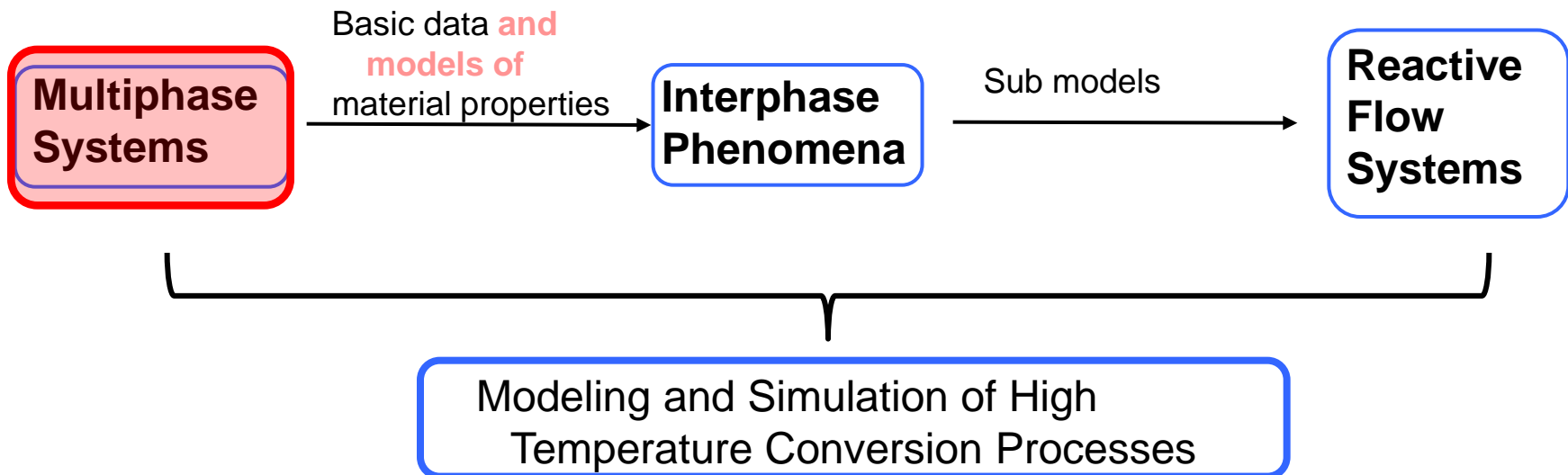
# Thermodynamic database development on the $\text{CaO-SiO}_2\text{-M}_2\text{O}$ (M= Na, K) systems

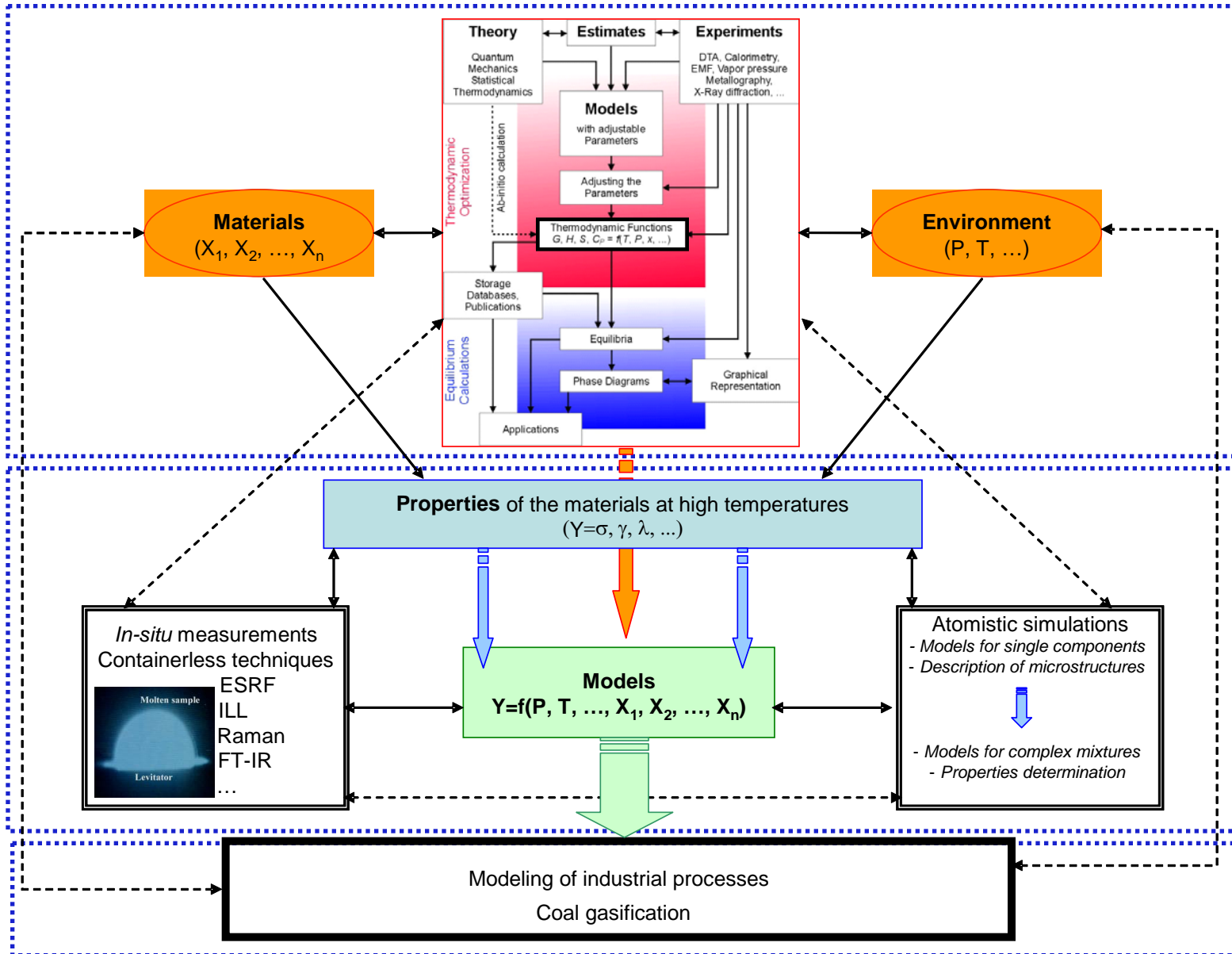
Ligang Zhang, Stefan Guhl



- **Motivation**
- **Calorimetry determination on  $C_p$  of intermediate phases in  $M_2O-CaO-SiO_2$  ( $M= K, Na$ ) ternary systems**
- **Ab initio calculations on the intermediate phases**
- **Phase diagram and thermodynamic database of  $M_2O-CaO-SiO_2$  ( $M= K, Na$ ) ternary systems**

- acronym for **Virtual High Temperature Conversion**
- established in 2009 as **Centre of Innovation Competence**
- combined initiative of 8 institutes at **TU Freiberg**
- consists of three research groups (about 20 research associates)





THERMOCHEMICAL  
PROPERTIES

THERMOPHYSICAL  
PROPERTIES

**Theory**



**VASP**

Quantum chemistry, ab initio, thermodynamics

**Experiments**



DTA, DSC, Calorimetry

**Optimization  
(Thermo-calc)**

**Phase diagram**

**Application**

$$G^\phi = \sum_i x_i {}^0G_i^\phi + RT \sum_i x_i \ln(x_i) + {}^{ex}G^\phi$$

binary

Assessment:  $\sum {}^{ex}G_{bin}^\phi$

ternary

Extrapolation ( $\sum {}^{ex}G_{bin}^\phi$ )  
+ Assessment:  $\sum {}^{ex}G_{ter}^\phi$

quaternary

Extrapolation ( $\sum {}^{ex}G_{bin}^\phi + \sum {}^{ex}G_{ter}^\phi$ )  
+ Assessment:  $\sum {}^{ex}G_{qua}^\phi$

as necessary

$${}^E G_m = \sum_{v=0}^{v=s} y_i y_j y_k \sum_{v=0}^{v=n} {}^v L_{i,j,k} (y_j - y_k)^v + \sum_{v=0}^{v=n} y_i y_j y_k y_l {}^v L_{i,j,k,l} (y_i - y_j)(y_k - y_l)$$

## Part I

# Cp of intermediate phases

**Theory**



**VASP**

Quantum chemistry, ab initio, thermodynamics

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DTA, DSC, Calorimetry

**Optimization**

**Phase diagram**

**Application**

$$G^\phi = \sum_i x_i {}^0G_i^\phi + RT \sum_i x_i \ln(x_i) + {}^{ex}G^\phi$$

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Extrapolation ( $\sum {}^{ex}G_{bin}^\phi + \sum {}^{ex}G_{ter}^\phi$ )

+ Assessment:  $\sum {}^{ex}G_{qua}^\phi$

as necessary



## Cp of intermediate phases

- Setaram AlexSys 1000 Calvet Calorimeter
- Setaram MHTC96 Calorimeter



Drop Calorimetry:  
calibration with sapphire pieces (NIST)  
operation under Air  
three to four drops per T-step

$$\Delta_{rel}H = \int_{T_{reservoir}}^{T_{furnace}} Cp(T) dT$$



$$\Delta_{rel}H = f(T)$$



$$\frac{\partial \Delta_{rel}H(T)}{\partial T} = Cp(T)$$



Single phase samples were prepared by Dr. V. Kahlenberg, Univ. Innsbruck



Preparation via co-precipitation

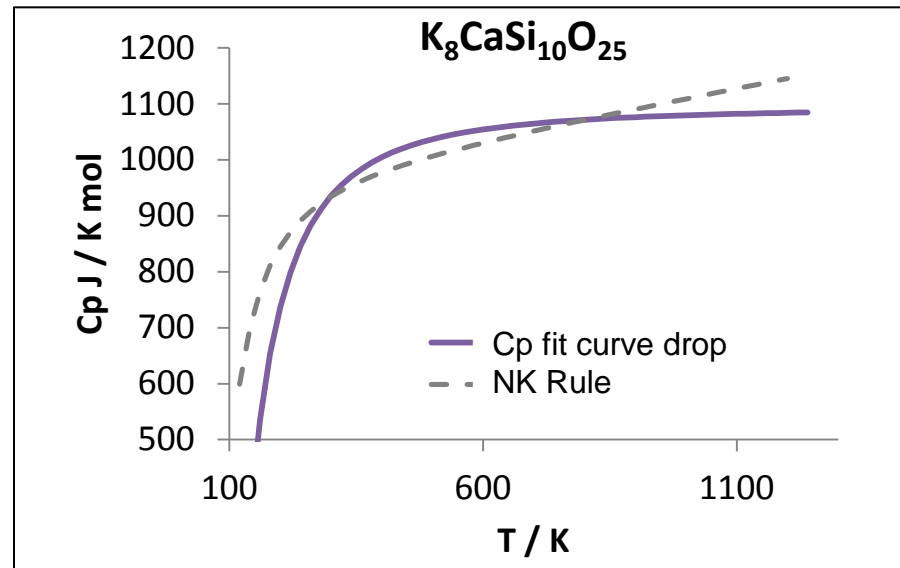
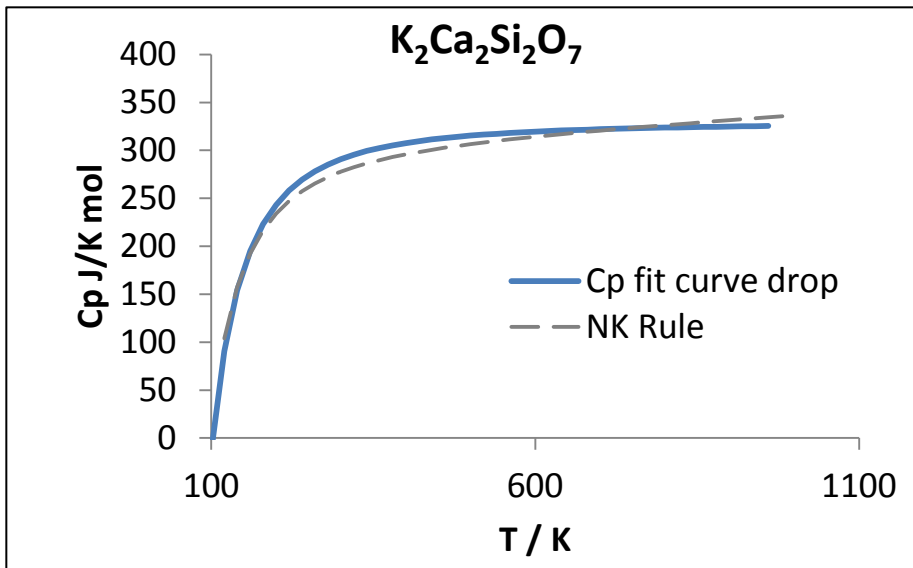
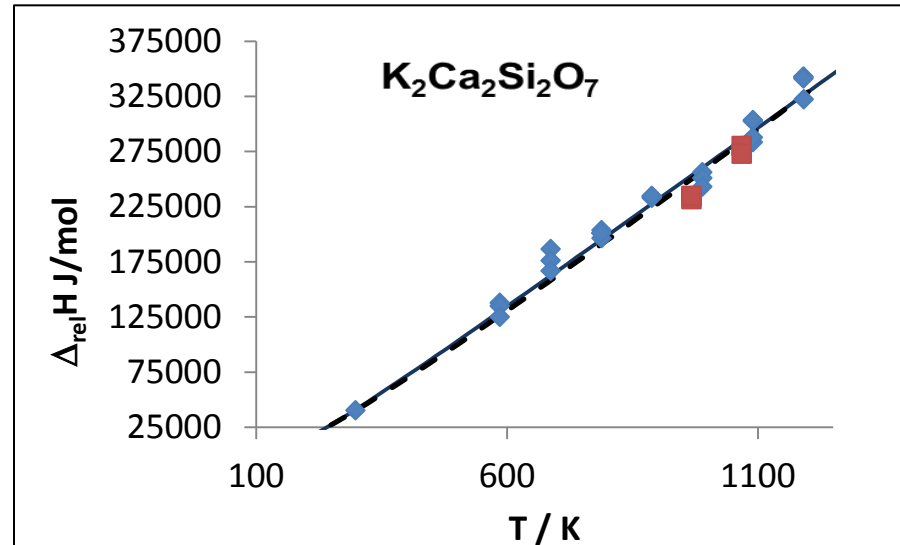
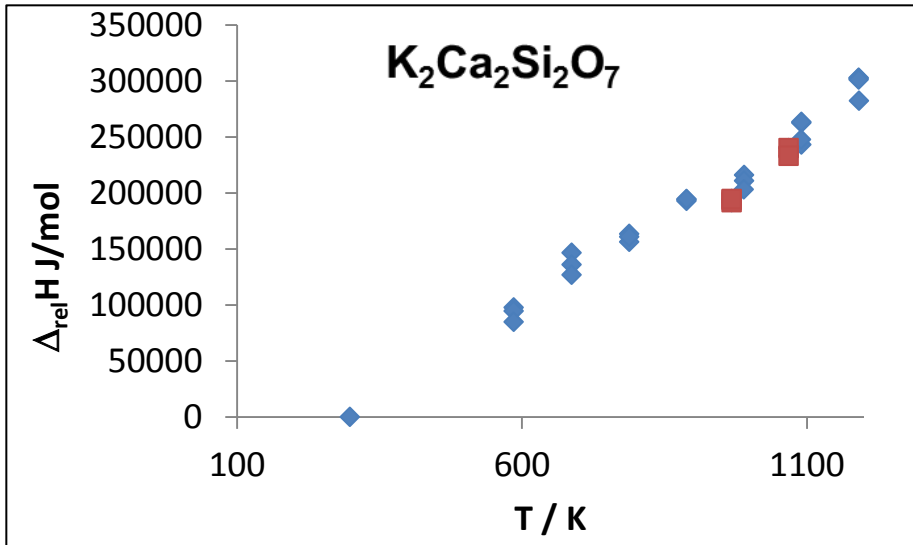
Supplied as powders

Checked for presence of other phases (XRD)

- Samples are sensitive to environment, especially moisture  
*use immediately*  
*store in desiccator*
- Powders not suitable for drop calorimetry  
*press sample pellets*
- Pellets are fragile  
*careful handling during weighing and actual drop*

**... experiments are time consuming and delicate**

# Calorimetry - Drop model



## Cp functions; Cp in J/K mol



$$C_p = 329.1 - 3.43 \cdot 10^6 \cdot T^{-2} - 6.59 \cdot 10^{-15} \cdot T^{-0.5}$$



$$C_p = 329.1 - 1.73 \cdot 10^{-13} \cdot T^{-2} - 5.6475 \cdot 10^3 \cdot T^{-0.5}$$



$$C_p = 1094 - 1.43 \cdot 10^7 \cdot T^{-2}$$

## • Setaram MHTC96 Calorimeter

HF-DSC measurement:  
 Standard with sapphire pieces (NIST)  
 operation under air  
 300-1200 °C



A set of three measurements is required, each with a different substance in the measurement crucible:

1. empty crucible (the “base line”)
2. standard
3. sample

$$C_{PS} = \frac{m_s \cdot M_{Std}}{M_s \cdot m_{Std}} \cdot \frac{S_S - S_{BL}}{S_{Std} - S_{BL}} \cdot C_{p_{Std}}$$

**Quantities:**

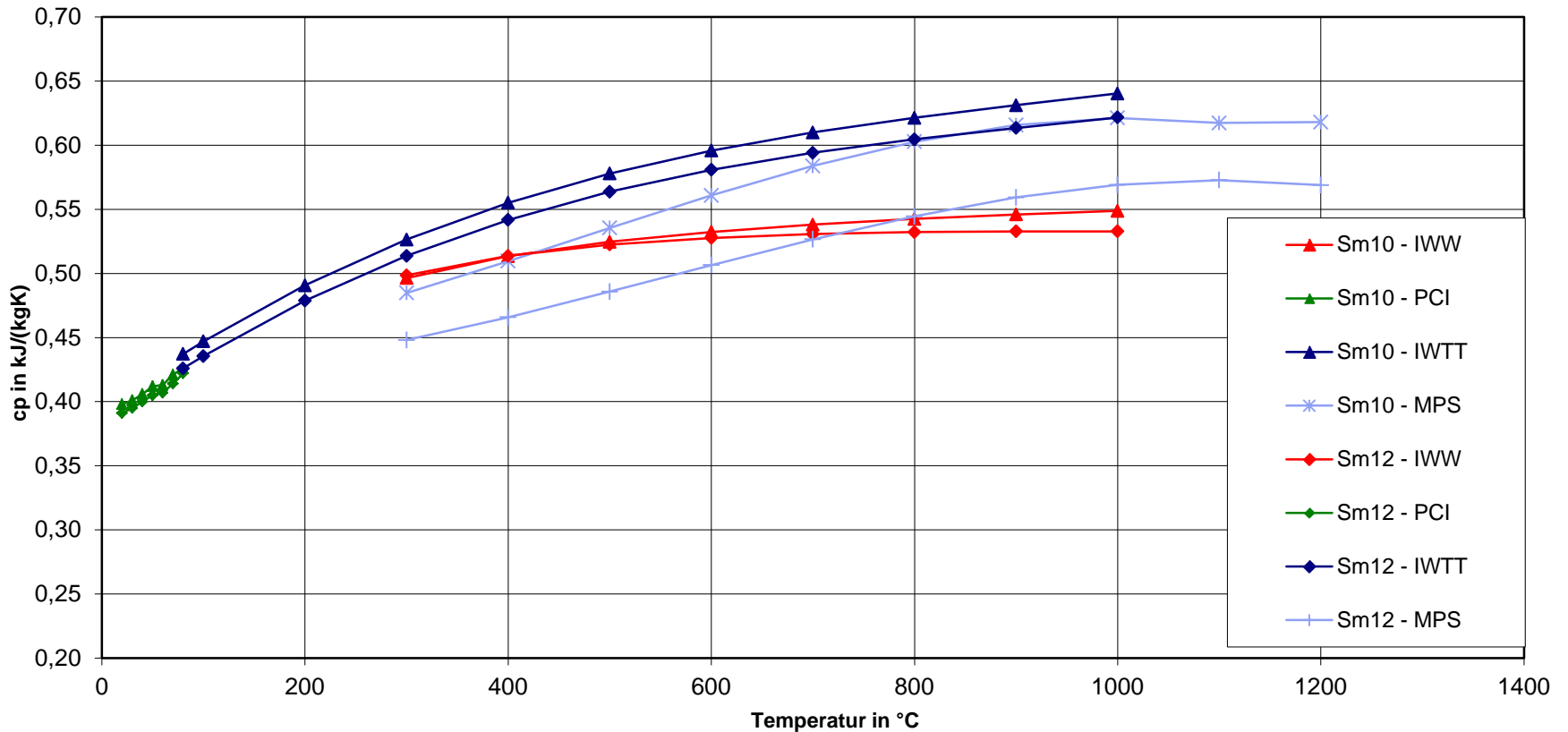
m ... mass (mg)    M ... molar mass (g/mol)    S ... DSC signal (µV)

**Indices:**

S ... sample    Std ... standard    BL ... base line

- Samples and Participants:
  - SM10: ZrO<sub>2</sub>-40mol%, SmO<sub>1.5</sub>-40mol% and YO<sub>1.5</sub>-20mol%
  - SM12: ZrO<sub>2</sub>-45mol%, SmO<sub>1.5</sub>-45mol% and YO<sub>1.5</sub>-10mol%

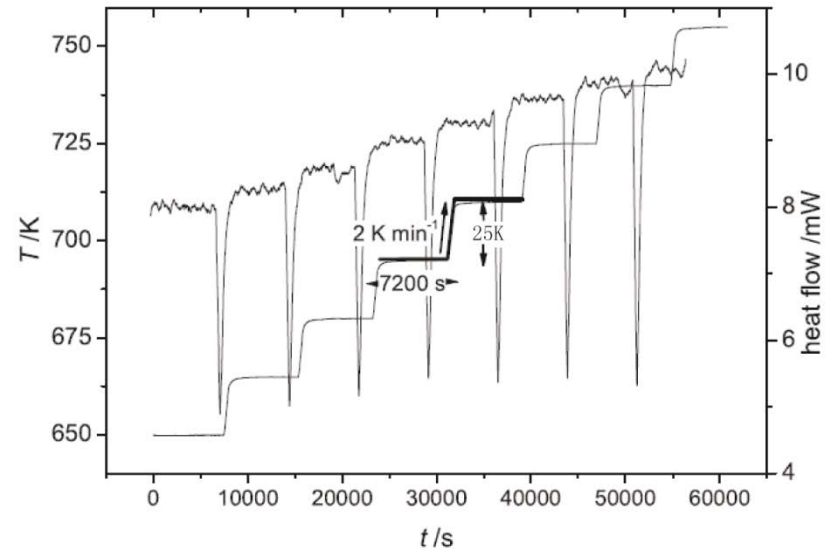
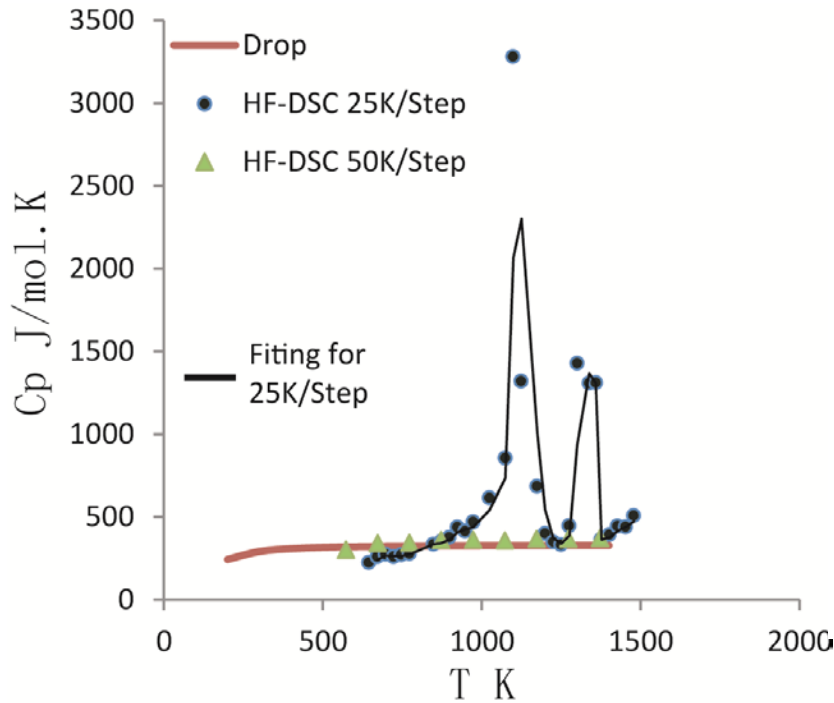
Group	Equipment	Sample	Atomosphere	Mode
VTC-MPS	SETARAM MHTC 96	Powder	Air	Stepwise: 25K/Step 2K/min Stability for 2 Hours
PCI-Institut für Physikalische Chemie (Dr. Seidel)				
IWTT-Institut für Wärmetechnik und Thermodynamik (Frau Dr. Wulf)	SETARAM MHTC 96	Powder	Air	Continuous Method: 3K/min
IWW-Institut für Werkstoffwissen- schaft (Herr Kriegel)	SETARAM TG-DTA	Powder	Ar	Continuous Method: 20K/min



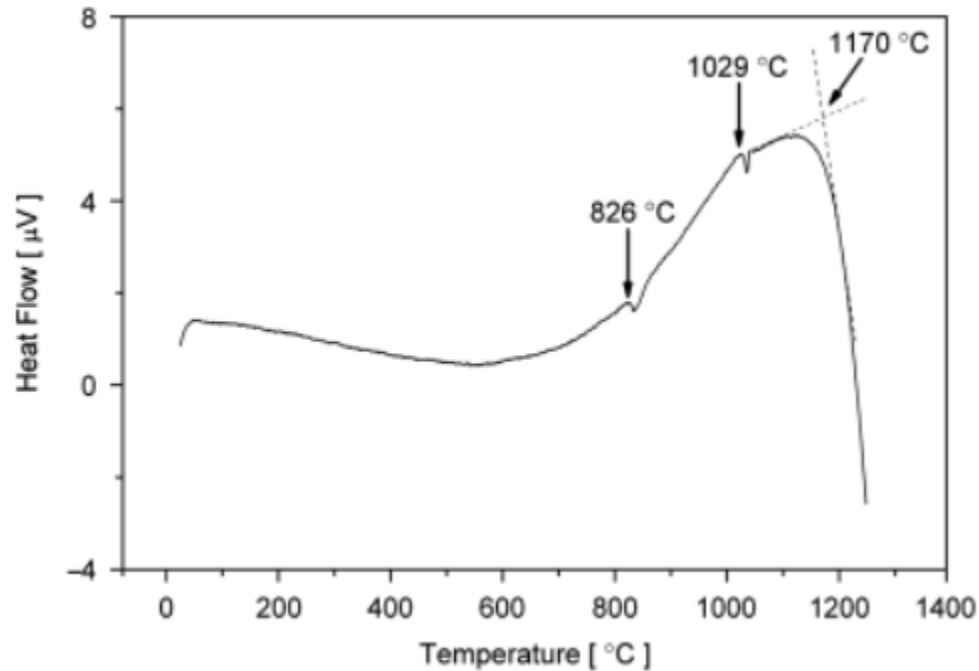
- Results within scattering of calorimetric measurements ( 0.15-0.20 KJ Kg<sup>-1</sup> K<sup>-1</sup> [1])
- Application of DSC mode beside Drop method within MPS (T-resolution, time saving)

[1] O. Fabrichnaya et al.: Calorimetric investigation of the La<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>, Nd<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>, Sm<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub> and LaYO<sub>3</sub> compounds and CALPHAD assessment of the La<sub>2</sub>O<sub>3</sub>–Y<sub>2</sub>O<sub>3</sub> system. Thermochimica Acta 526, 2011

- Application of DSC-mode and comparison to Drop mode for  $K_2Ca_2Si_2O_7$
- DSC-mode with 50 K- and 25 K steps, 1-2 weeks



- Application of DSC-mode and comparison to Drop mode for  $K_2Ca_2Si_2O_7$
- Comparison to DTA-data from literature [1]:



- Repetition measurement
- XRF analysis?

[1] E. Arroyabe, R. Tessadri, D. M. Tobbens, V. Kahlenberg, J. American Ceramic Society, 98, 2652-2655



# Part II

## VASP (Vienna Ab initio Simulation Package)

**Theory**



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Quantum chemistry, ab initio, thermodynamics

**Experiments**



DTA, DSC, Calorimetry

**Optimization**

**Phase diagram**

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as necessary  
as necessary

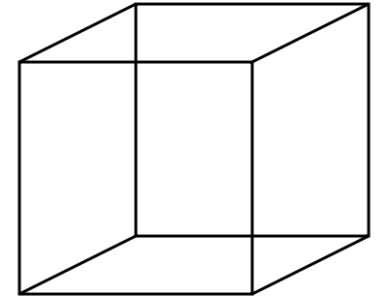
## Structure detail:

Structure: FCC, BCC, HCP ...

Kinds of atoms

How many atoms in one cell

Atoms occupied sites



## System studied (input file) :

INCAR (how to do)

POSCAR (position + car)

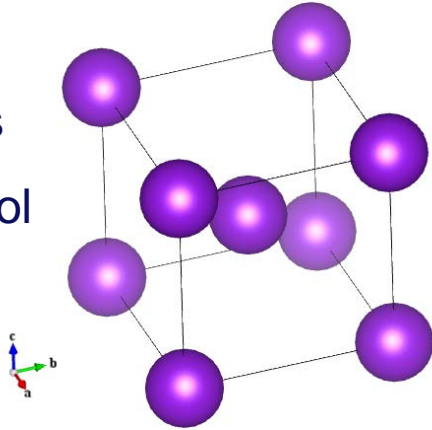
POTCAR (Pseudo-potential)

KPOINTS (how many steps)

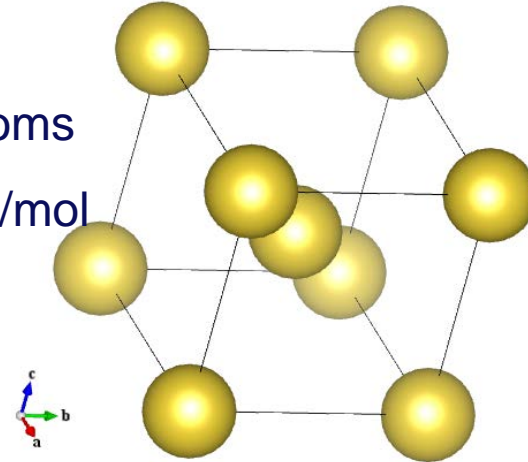
# Pure elements

**Calculated Enthalpy at 0K**  $H_f^0$

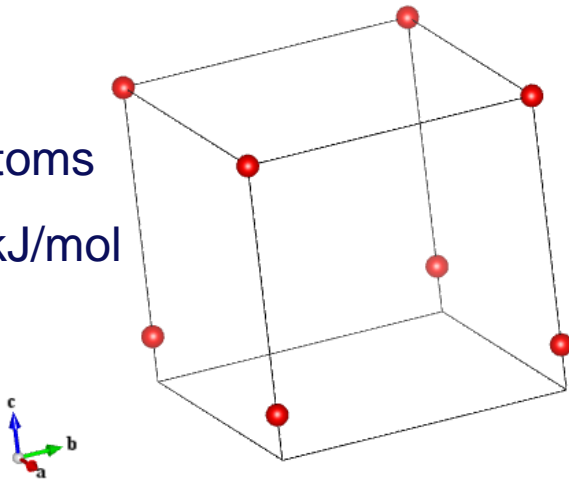
K: 2 atoms  
-113.3 kJ/mol



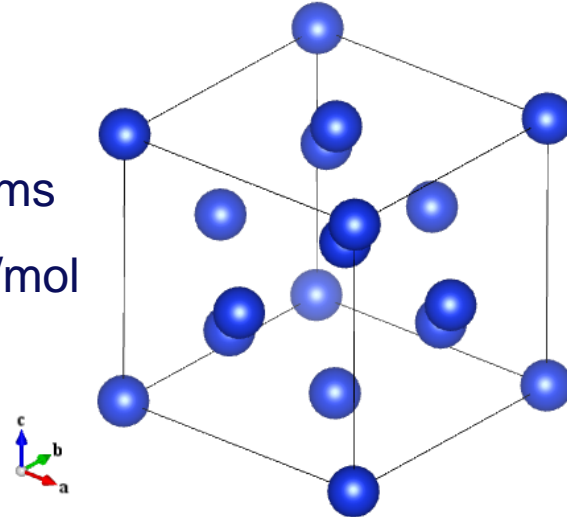
Na: 2 atoms  
-138.8 kJ/mol



O: 2 atoms  
-475.7 kJ/mol



Si: 8 atoms  
-573.8 kJ/mol



K<sub>2</sub>SiO<sub>3</sub>: 72 atoms

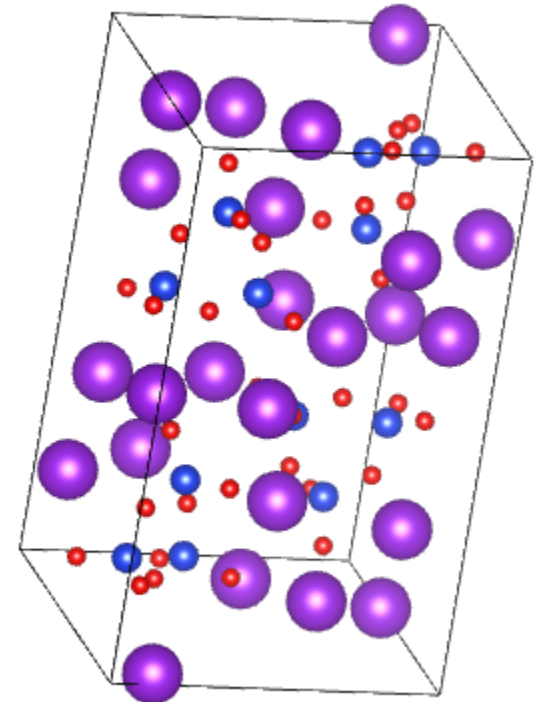
-643.11 kJ/mol

Reference state: K, Si and O

$$H_f^0 = -1634 \text{ kJ/mol}$$

NIST-JANAF data :  $H_f^{298} = -1548.08 \text{ kJ/mol}$

In CALPHAD approach: the reference state should be: K<sub>2</sub>O and SiO<sub>2</sub>



The enthalpy is -63.9 kJ/mol

From our own optimization [1] :

$$H_f^{298} = -61.3 \text{ kJ/mol}$$

[1] L. Zhang, C. Schmetterer, P. Masset, Computational Materials Science, In press 2012

# Comparing the VASP with CALPHAD results

Compounds	K <sub>2</sub> SiO <sub>3</sub>	K <sub>2</sub> Si <sub>2</sub> O <sub>5</sub>	K <sub>2</sub> Si <sub>4</sub> O <sub>9</sub>	Na <sub>2</sub> SiO <sub>3</sub>	Na <sub>4</sub> SiO <sub>4</sub>	Na <sub>2</sub> Si <sub>2</sub> O <sub>5</sub>	Na <sub>6</sub> Si <sub>8</sub> O <sub>19</sub>
Atoms	72	108	30	24	18	36	136
Reference state are K <sub>2</sub> O, Na <sub>2</sub> O and SiO <sub>2</sub>							
Enthalpies of formation kJ/mol							
VASP At 0K	-63.9	-55.8	-50.4	-103.7	-106.4	-68.4	-88.1
CALPHAD At 298K	-61.3	-55.0	-49.9	-95.8	-104.5	-67.4	-61.7

- As the limited computational resources, we just using quasi-harmonic Debye model to calculate the Cp.

$$E(V) \implies B_S \simeq B_{\text{static}}(\mathbf{x}) \simeq B_{\text{static}}(\mathbf{x}_{\text{opt}}(V)) = V \left( \frac{d^2 E(\mathbf{x}_{\text{opt}}(V))}{dV^2} \right) = V \left( \frac{d^2 E(V)}{dV^2} \right)$$

$$\Theta = \frac{\hbar}{k} [6\pi^2 V^{1/2} n]^{1/3} f(\sigma) \sqrt{\frac{B_S}{M}}, \quad f(\sigma) = \left\{ 3 \left[ 2 \left( \frac{2}{3} \frac{1+\sigma}{1-2\sigma} \right)^{3/2} + \left( \frac{1}{3} \frac{1+\sigma}{1-\sigma} \right)^{3/2} \right]^{-1} \right\}^{1/3}$$



$$C_{v,\text{vib}} = 3nk \left[ 4D(\Theta/T) - \frac{3\Theta/T}{e^{\Theta/T} - 1} \right]$$

$$D(y) = \frac{3}{y^3} \int_0^y \frac{x^3}{e^x - 1} dx$$

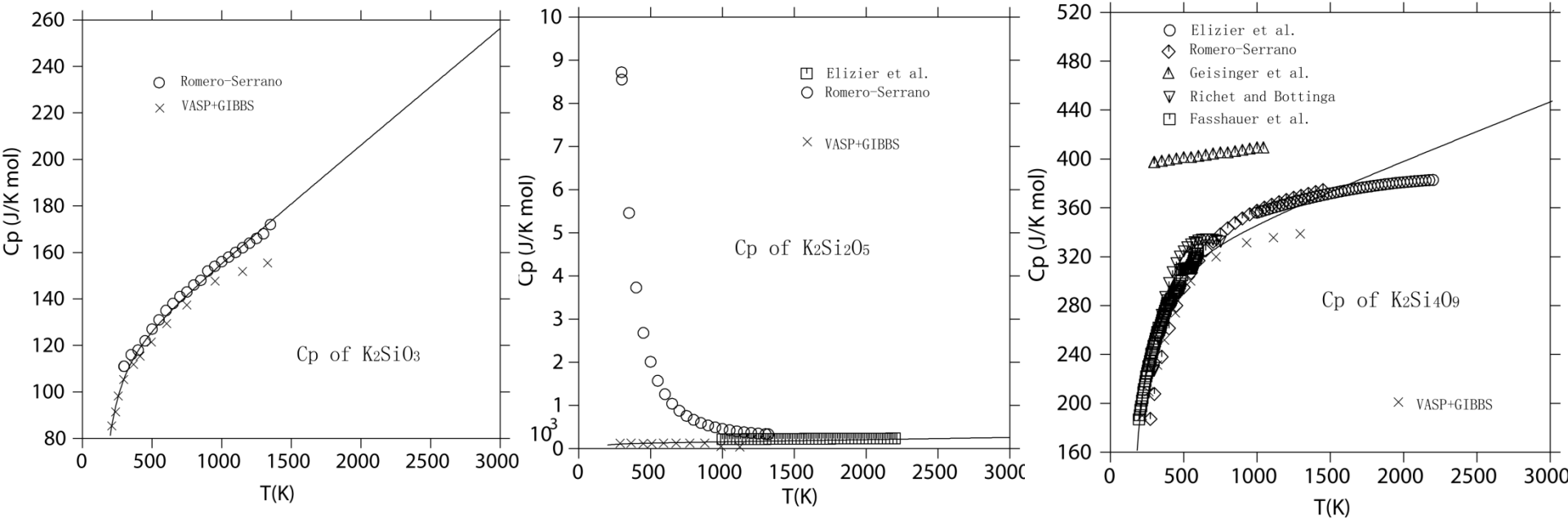
$$\gamma = - \frac{d \ln \Theta(V)}{d \ln V} \implies \alpha = \frac{\gamma C_{v,\text{vib}}}{B_T V}$$

$$C_{p,\text{vib}} = C_{v,\text{vib}} (1 + \alpha \gamma T)$$

Reference:

M.A. Blanco et.al., Computer Physics Communications, Vol 158,2004,P57-72

Here show some results on  $K_2O-SiO_2$  system from VASP+GIBBS comparing to our CALPHAD results and literature data



We can easily find that  $C_p$  from VASP+GIBBS is in good agreement with both CALPHAD and literature results, however, when it come to high temperature ( $>1000K$ ), the results are more and more negative compare to CALPHAD results, that is because both VASP and GIBBS are more reliable in low temperature range.

That means we still need experimental measurements.



## Part III

# CALPHAD optimization

**Theory**

**Experiments**

**VASP**

Quantum chemistry, ab initio, thermodynamics

DTA, DSC, Calorimetry

**Optimization**

**Phase diagram**

**Application**

$$G^\phi = \sum_i x_i {}^0G_i^\phi + RT \sum_i x_i \ln(x_i) + {}^{ex}G^\phi$$

binary

Assessment:  $\sum {}^{ex}G_{bin}^\phi$

ternary

Extrapolation ( $\sum {}^{ex}G_{bin}^\phi$ )  
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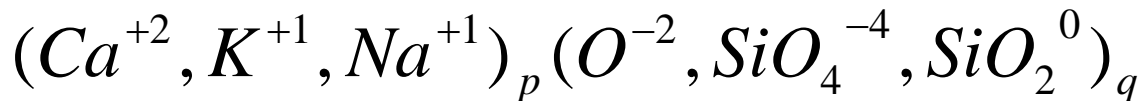
quaternary

Extrapolation ( $\sum {}^{ex}G_{bin}^\phi + \sum {}^{ex}G_{ter}^\phi$ )  
+ Assessment:  $\sum {}^{ex}G_{qua}^\phi$

as necessary

Model of Liquid phase:

Two sublattice model for the ionic liquid phase (the Gibbs energy is extrapolated from the three binary systems)



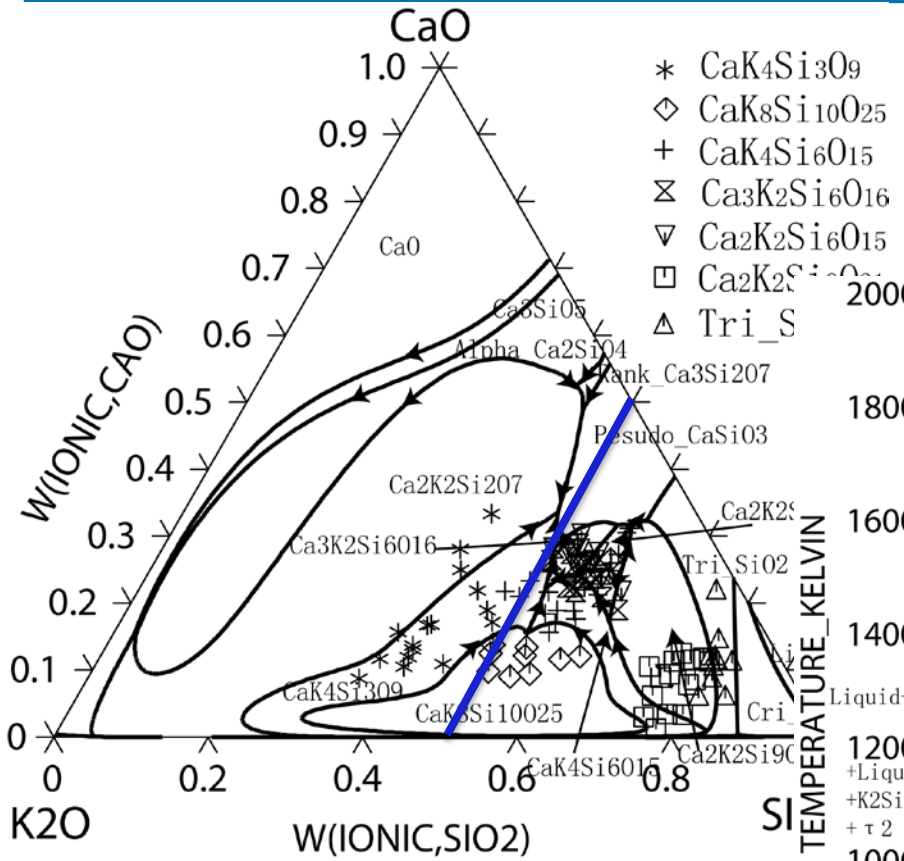
Model of solid phases:

All of the Gibbs energy values (7 ternary compounds) are given relative to enthalpy of selected reference states for the elements at 298.15K. This state is denoted by SER (Stable Element Reference).

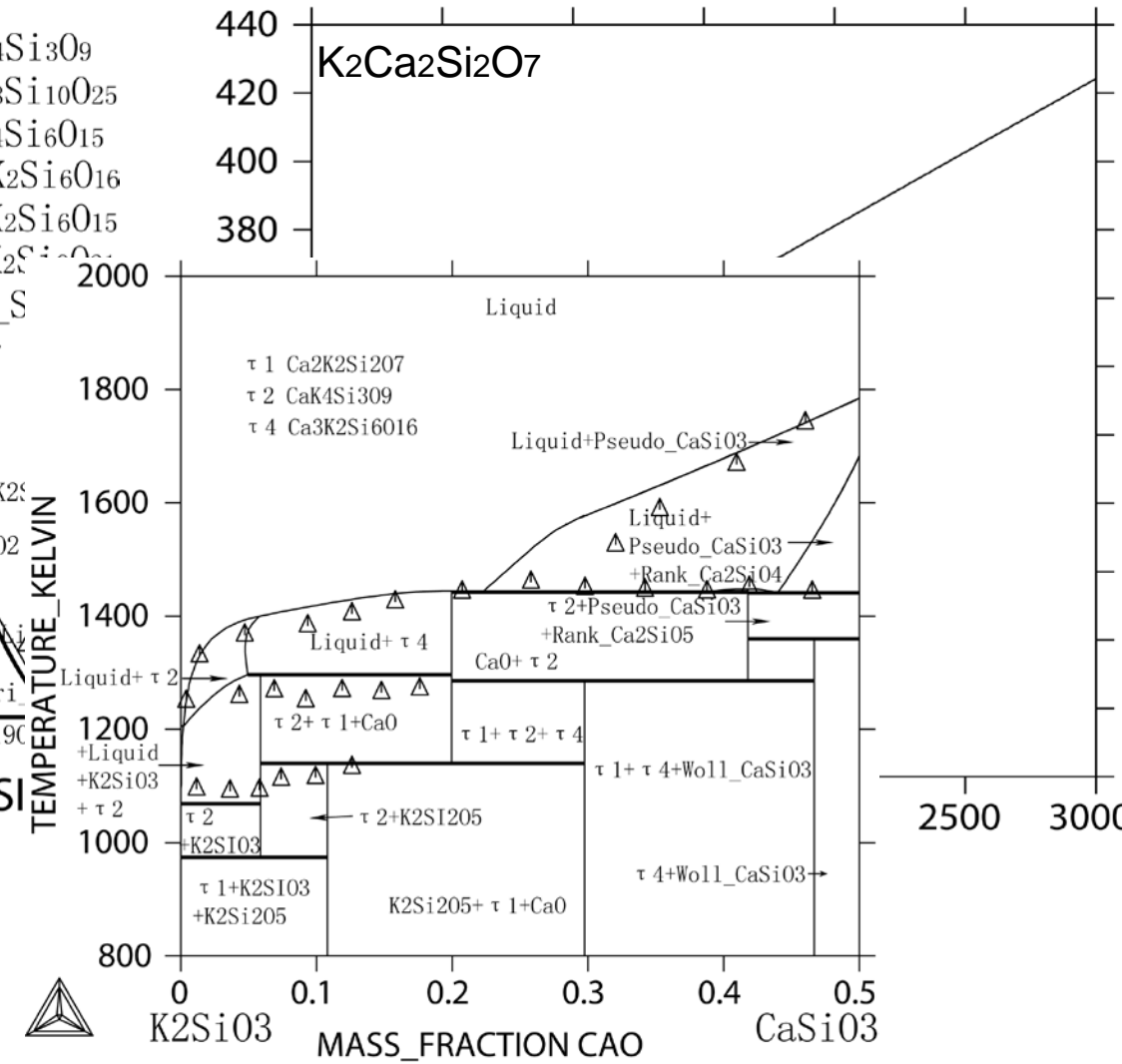
$${}^0G - H^{SER} = A + B/T + CT + DT \ln T + ET^2$$

Where the parameters **A**, **B**, **C**, **D**, **E** should be optimized in this work

# CaO-SiO<sub>2</sub>-K<sub>2</sub>O ternary system



- \* CaK<sub>4</sub>Si<sub>3</sub>O<sub>9</sub>
- ◇ CaK<sub>8</sub>Si<sub>10</sub>O<sub>25</sub>
- + CaK<sub>4</sub>Si<sub>6</sub>O<sub>15</sub>
- ⊗ Ca<sub>3</sub>K<sub>2</sub>Si<sub>6</sub>O<sub>16</sub>
- ▽ Ca<sub>2</sub>K<sub>2</sub>Si<sub>6</sub>O<sub>15</sub>
- Ca<sub>2</sub>K<sub>2</sub>Si<sub>9</sub>O<sub>19</sub>
- △ Tri\_S



Calculated thermodynamic of K<sub>2</sub>O-CaO-SiO<sub>2</sub> system

## Database Development

### Cooperation with GTT-Technologies:

#### GTT oxide Database

- developed since 2009
- Components:  $\text{SiO}_2$ ,  $\text{CaO}$ ,  $\text{MgO}$ ,  $\text{K}_2\text{O}$ ,  $\text{Na}_2\text{O}$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{FeO}$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{CrO}$ ,  $\text{Cr}_2\text{O}_3$ ,  $\text{P}_2\text{O}_5$ ,  $\text{CaS}$ ,  $\text{FeS}$ ,  $\text{MgS}$  and  $\text{CaF}_2$
- consistent treatment of the alkali components, improved modelling of spinel, mullite, mellilite and several other solid solutions
- non-ideal associate model for liquid

#### VTC-MPS

- $\text{M}_2\text{O-SiO}_2\text{-CaO-Al}_2\text{O}_3$  systems (M= K, Na)
- Thermodynamic Data ( $H_f$ ,  $c_p$ ) from calorimetric measurement and ab initio calculations
- G(T)-functions of optimized subsystems by CALPHAD method

#### FZ IEK-2

Synthesis of ternary compounds for calorimetry

- Integration of VTC-MPS DATA in GTT oxide Database by GTT Technologies
- Application of Database for the estimation of the ash/ slag behaviour within VTC

- Datapool MPS

System	Phases	Gibbs energy function (CALPHAD)	Enthalpy of formation (VASP)	Heat capacity (VASP+GIBBS)	Heat capacity (EXP.)
CaO-SiO <sub>2</sub>	1	1			1 (Drop mode)
K <sub>2</sub> O-SiO <sub>2</sub>	3	3	3	3	
Na <sub>2</sub> O-SiO <sub>2</sub>	8	8	3	3	
Al <sub>2</sub> O <sub>3</sub> -K <sub>2</sub> O	3	(3)	1	[1]	
Al <sub>2</sub> O <sub>3</sub> -Na <sub>2</sub> O	3	(3)	1	[1]	
CaO-K <sub>2</sub> O-SiO <sub>2</sub>	7	7	(2)	(2)	3 (Drop mode) 1 (Stepwise mode)
CaO-Na <sub>2</sub> O-SiO <sub>2</sub>	6	6	[4]	[4]	2[Drop mode]
Al <sub>2</sub> O <sub>3</sub> -K <sub>2</sub> O-SiO <sub>2</sub>	2	[2]			
Al <sub>2</sub> O <sub>3</sub> -Na <sub>2</sub> O-SiO <sub>2</sub>	2	[2]			

- () = in work
- [] = planed

In this work, CALPHAD optimization of ternary phase diagrams were reached based on experimental and ab-initio data

*...work in  
progress...*

There is still a lot do ...

... Heat capacities of ternary system need to be calculated

... experimental Cp need to repeat

**Thanks to:** Colleagues from VIRTUHCON

Prof. V. Kahlenberg (Univ. Innsbruck)

For financial support:



Thank you for your attention