





Thermodynamics of Lithium Battery Materials

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Gunnar Eriksson Symposium 2012

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Thermodynamics of Lithium Battery Materials

- Introduction
- Why battery thermodynamics and phase diagrams? Calculation of voltages, ...
- Calorimetry of lithium ion batteries
 - **Isothermal and adiabatic calorimetry**
- Li Mn O and Li Fe P O materials systems

Gunnar Eriksson Symposium 2012







Electric vehicles - some examples



Mini E (L.A., NY, Berlin; BMW / Vattenfall) Operating range 250 km

Smart electric drive (London, Berlin; Daimler / RWE) Operating range 135 km







Electric vehicles - some examples



i MiEV, Mitsubishi

Capacity of Battery: 16 kWh (200kg) 150 km operating range

Per single cell: Energy density 218 Wh/I and Specific energy 109 Wh/kg; Specific power 550 W/kg Total of 176 Cells in Battery System







Energy stored in 100 kg of best commercial batteries good enough for just 100 km operating range

Batteries for Electric Vehicles have to be optimized to high energy density (kWh), save thermal behavior and cycle life

Energy density (Wh/I) and specific energy (Wh/kg) determine for a given operating range the volume and mass of battery





(2) FreedomCAR and Fuel Partnership and United States Advanced Battery Consortium (USABC), Electrochemical Energy Storage Technical Team Technology Development Roadmap (Southfield, MI: USCAR, 2006) Thermodynamics and phase diagrams govern battery performance

Thermal management of batteries (Heat dissipation and temperature distribution)

Battery safety (thermal runaway)

Structural changes in active materials

Surface energies (nano-micro)

Synthesis of complex ceramic materials









Presently: Anode: graphitic C; Electrolyte: LiPF₆ in organic carbonate solvents; Cathode: Co-based LiMO₂



German Research Foundation, Priority Programme 1473 Materials with New Design for Improved Lithium Ion Batteries -WeNDeLIB

DFG Schwerpunktprogramm 1473, Kick-Off-Treffen, Werkstoffe mit neuem Design für verbesserteLithium-Ionen-Batterien - WeNDeLIB

Started in October 2010

Program Commission:

Hans Jürgen Seifert Rainer Schmid-Fetzer Martin Winter Nicola Hüsing Andreas Gutsch Bengt Hallstedt Ingo Steinbach Karlsruher Institut für Technologie (Chair) Technische Universität Clausthal (Co-Chair) Universität Münster (Co-Chair) Universität Salzburg Li-Tec, Kamenz, now with KIT RWTH Aachen ICAMS, Bochum

International Member:

Alexandra Navrotsky University of California, Davis

13 joint projects (36 projects in total) + coordinator project

German Research Foundation, Priority Programme 1473 Materials with New Design for Improved Lithium Ion Batteries -WeNDeLIB

Vision of this priority program: Coupling of materials thermodynamics and kinetics and modelling of materials design for innovative lithium ion batteries

Ideal case:

Complete design of battery before the experimental work or processing starts in the lab or in industrial operation

Still a long way to go - We introduce a new *interdisciplinary* concept for LiB (no international program to be compared with this approach)





Approach by electrochemical cell reaction equations



Write reaction as

 $\mathbf{x} \mathbf{A} + \mathbf{A}_{\mathbf{y}} \mathbf{R} = \mathbf{A}_{\mathbf{y} + \mathbf{x}} \mathbf{R}$

 $\Delta G_{\Gamma}^{\circ}$ = Gibbs free energy change due to the reaction

= $\Sigma \Delta G f^{\circ}(\text{products}) - \Sigma \Delta G f^{\circ}(\text{reactants})$

This can be related to the cell voltage

$$\Delta \mathbf{Gr}^{\circ} = -\mathbf{z} \mathbf{q} \mathbf{E}$$

- z = charge number carried by ions in electrolyte
- q = elementary charge per particle
- E = externally measurable cell voltage

Approach by electrochemical cell reaction equations



If $\Delta G_{\Gamma^{\circ}}$ has units of energy/mol, then

 $\Delta \mathbf{G}_{\mathbf{I}^{\circ}} = -\mathbf{Z} \mathbf{F} \mathbf{E}$

F = charge per mol = 96,500 Coulombs/mol



Approach by phase diagrams

Coulometric Titration of Li Into Al at 400°C



Measured Using LiCl-KCl as a Molten Salt Electrolyte

C.J. Wen, B.A. Boukamp, R.A. Huggins, W. Weppner, J. Electrochem. Soc. <u>126</u>, 2258 (1979)



Electrochemical conversion mechanism



$$CuO + 2 \cdot e^- + 2 \cdot Li^+ \longleftrightarrow Cu^0 + Li_2O$$

- Cu-Fe-O electrodes for LIB exhibit
 - Fe-oxides: high theoretical capacity
 - Cu-oxides: cycling stability

Calculated Li-Cu-O system (CALPHAD result); isothermal section







More relationships Thermodynamics - Phase Diagrams - Electrochemistry



 $Li_xAN + CA \xrightarrow{\text{discharge}} AN + Li_xCA$

Full lithium ion cell discharge reaction
$$Li_xAN + CA \xrightarrow{\text{discharge}} AN + Li_xCA$$
 $E_0(x,T)$ Open circuit voltageFree energy of the full reaction is: $\Delta G(x,T) = -n \cdot F \cdot E_0(x,T)$ F Free energy of the full reaction can be written: $\Delta G(x,T) = -\Delta H(x,T) - T\Delta S(x,T)$ F Neglecting T-dependence: $\Delta G_0(x,T) = -\Delta H(x) - T\Delta S(x)$ ΔH $\Delta G_0(x,T) = -\Delta H(x) - T\Delta S(x)$ Temperature slope of $E_0(x,T)$ Combining equations: $\Delta H(x) = F\left(-E_0(x,T) + T\frac{\partial E_0(x,T)}{\partial T}\Big|_x\right)$

In-situ technique "entropymetry" and phase diagrams



x in Li_{COO}

Batteries, WILEY-VCH (2010)

Reversible and irreversible heat during cell reaction

$$Q_r = T\Delta S \frac{I}{nF}$$
 Reversible heat generation

with

$$\Delta S(x) = F\left(\frac{\partial E_0(x,T)}{\partial T}\Big|_x\right)$$

 $Q_{irr} = -I^2 R_i$ Irreversible heat generation

$$Q = Q_r + Q_{irr}$$
 Total heat generation

+ side reactions



40Ah pouch cell





B. Ketterer, U. Karl, D. Möst, S. Ulrich, Wissenschaftliche Berichte FZKA 7503 (2009)

40Ah cell with eight thermocouples measured in Accelerating Rate Calorimeter (ARC)







Isothermal cycling of a 40Ah pouch cell Discharge parameter: 4 method: constant current Current (A) 2 1.5 0.9 (CC)SSF 0 Current / mA 0.0 0.0 $U_{min} = 3.0V$ -2 $I = 5A \rightarrow C/8$ -rate -4 -0.3 Charge Charge Discharge Discharge Charge parameter: -0.6 method: constant current, 3.0 5.0 5.5 4,0 2.5 3.5 4.5 4.0 Voltage / V constant voltage (CCCV) Voltage (V) $U_{max} = 4,1V$ 3,5 $I = 5A \rightarrow C/8$ -rate $I_{min} = 0.5A$ 3,0 Charge Discharge Discharge Charge at room temperature ∆T=1,431K ∆T=1,353K 26,5 Temperature (°C) ∆T=-1,437K ∆T=-1,419K temperature coefficient 26,0 < 0 25,5 25,0 0 500 1000 1500 2000 Procedure like in: Time (min) H.-B. Ren, et. al., Int. J. Electrochem. Sci. 6, p.

727 – 738 (2011)

A.K. Shukla, T.P. Kumar, Current Sci. 94, p. 314-331 (2008)

Temperature measurements on a 40Ah pouch cell







Adiabatic cycling of a 40Ah pouch cell Discharge parameter: method: constant current (A) 1.2 2 Current 0,9 (CC)SSF 0 Current / mA 0 :0 $U_{min} = 3.0V$ -2 $I = 5A \rightarrow C/8$ -rate -4 -0.3 Charge Discharge Charge Discharge Charge parameter: -0.6 method: constant current, 4,0 5.0 5.5 2.5 3.0 3.5 4.5 4.0 Voltage / V constant voltage (CCCV) Voltage (V) $U_{max} = 4,1V$ 3,5 $I = 5A \rightarrow C/8$ -rate $I_{min} = 0.5A$ 3,0 Discharge Charge Discharge Charge 40 at room temperature ∆T=4,932K Temperature (°C) 35 ∆T=2,928K temperature coefficient ∆T=1,252K ∆T=6,537K 30 < 025 500 1000 0 1500 2000 Procedure like in: Time (min)

H.-B. Ren, et. al., Int. J. Electrochem. Sci. 6, p. 727 – 738 (2011)

A.K. Shukla, T.P. Kumar, Current Sci. 94, p. 314-331 (2008)



Temperature measurements on a 40Ah pouch cell



Thermodynamically stable phases in the Li-Mn-O system.

Stoichiometric Cation-mixed

Mn-rich Mn-deficient $\begin{array}{l} \text{LiMn}_{2}\text{O}_{4} \text{ (c)} \\ \text{Li}_{1+x}\text{Mn}_{2-x}\text{O}_{4} \text{ (c)}; \ 0 < x \leq 1/3 \\ \text{Li}_{4}\text{Mn}_{5}\text{O}_{12} \\ \text{Li}_{1-x}\text{Mn}_{2+x}\text{O}_{4} \text{ (t)}; \ 0 < x < \sim 0.1 \\ \text{Li}_{1}\text{Mn}_{2-x}\text{O}_{4} \text{ (c)}; \ x = 1/4 \end{array}$



Thermodynamically unstable phases in the Li-Mn-O system.

De-lithiated Lithiated Oxygen-deficient Mn-deficient Oxygen-rich Cation-deficient $\begin{array}{l} \text{Li}_{1-y}\text{Mn}_{2}\text{O}_{4}\\ \text{Li}_{2}\text{Mn}_{2}\text{O}_{4}\ (t)\\ \text{Li}_{1+x}\text{Mn}_{2-x}\text{O}_{4-\delta};\ \delta{>>}0\\ \text{Li}\text{Mn}_{2-x}\text{O}_{4}\ (c);\ 0< x< 1/4\\ \text{Li}_{2}\text{Mn}_{4}\text{O}_{9}\ (c)\\ \text{Li}_{1-x}\text{Mn}_{2-y}\text{O}_{4}\ (c)\end{array}$



R. Huggins, Advanced Batteries, Springer 2011



Phase relationships in the system $LiMnO_2 - Li_2MnO_3 - \lambda - MnO_2$ and subsystem $LiMn_2O_4 - Li_4Mn_5O_{12} - Li_2Mn_4O_9$ (Yonemura et al. 2004).



- Samples prepared using the solid state method
- Phase diagram investigated at pO₂=0.21 atm

[1999 Pau] Chem. Mater., 11 (1999), 3065-3079.

p(O₂)=0.21 atm









[1996Gao] Gao and Dahn, J. Electrochem. Soc., 143(6), 1783-1788, 1996


Enthalpy of Drop Solution of $Li_{1+x}Mn_{2-x}O_{4-\delta}$

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Al2O3 plug	
SiO ₂ glass dropping tube for ———————————————————————————————————	
Platinum tube for bubbling gas	
Platinum crudible where solvent ——— is introduced	
SiO2 glass crucible	
SiO2glass liner	
Inconel protection tube —	- P
Sodium Molybdate	/ ,700°C

AlexSys, SETARAM High Temperature Calvet Calorimeter

Enthalpy of Drop Solution of $Li_{1+x}Mn_{2-x}O_{4-\delta}$



Sample Number	DROP 1	DROP 2	DROP 3	DROP 4	DROP 5	DROP 6	DROP 7	DROP 8
Date	08. Mai	09. Mai	09. Mai	09. Mai				
Mass pellet (mg)	6,00	5,14	6,10	6,62	4,85	5,35	4,83	5,59
T(room) (°C)	23,90	24,10	24,20	24,10	24,20	24,50	24,30	24,20
T(cal.) (°C)	700,40	700,40	700,40	700,40	700,40	700,40	700,40	700,40
Formula weight (g/mol)	180,815	180,815	180,815	180,815	180,815	180,815	180,815	180,815
Moles of LiMn2O4 (mol)	0,0000332	0,0000284	0,0000337	0,0000366	0,0000268	0,0000296	0,0000267	0,0000309
Peak Area [µV.s]	1832,4170	1538,5640	1765,6030	1910,2030	1402,3320	1544,6110	1397,2290	1611,2630
Calibration factor from Al ₂ O ₃ calibration[J/µV.s]	0,00462077	0,00462077	0,00462077	0,00462077	0,00462077	0,00462077	0,00462077	0,00462077
Measured Heat Effect (kJ/mol)	255,1652	250,0926	241,8308	241,0849	241,5781	241,2203	241,6957	240,8258
Accepted Measurement	0	0	1	1	1	1	1	1

Commercial cathode materials



Gravimetrical energy densities (capacities)



(b) $\mu_{O_1} = -11.52 \text{ eV}$, LiFePO₄ appears

Figure 2. Phase diagrams at less reducing environments.

Ong et al. (2008)

Li-Fe-P-O System



(a) $\mu_{O_2} = -11.74 \ eV$, Li₃Fe₂(PO₄)₃ disappears













Ong et al. (2008)



Figure S1. Calculated Li-Fe-P ternary phase diagram equilibrated with an oxygen potential under reducing conditions. The 'A' arrow indicates off-stoichiometry induced in the samples described in this Letter. The 'B' and 'C' arrow respectively indicate the more common one-to-one Fe/P deficiency and lithium deficiency. The shaded area in phase diagram indicates relevant coating compositions.

Kang and Ceder (2009)







Thermodynamics of Lithium Battery Materials

- Materials Thermodynamics, Phase Diagrams and Electrochemistry to be simultaneously investigated
- Fundamental for electrochemical cell performance, understanding of thermal and safety behavior and for preparation of cathode materials
- Thermodynamics and phase diagrams even for the most important systems e.g. Li-Co-O, Li-Mn-O and Li-Fe-P-O to be investigated in more detail

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Composition triangle in the system LiNiO_2 – LiMnO_2 – $\text{Li[Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$. Bold lines represent the compositions which have been reported as single phase via a high temperature solid state reaction Zhang et al. (2008).



(b)







 $3\text{LiMn}_2\text{O}_4 \leftrightarrow 3\text{LiMnO}_2 + \text{Mn}_3\text{O}_4 + \text{O}_2$

5.89% theoretical weight loss

Reaction of stoichiometric LiMn₂O₄ at Log(pO₂)=-2.3 not indicated in the critically evaluated potential diagram for stoichiometric LiMn₂O₄





 $3\text{LiMn}_2\text{O}_4 \leftrightarrow 3\text{LiMnO}_2 + \text{Mn}_3\text{O}_4 + \text{O}_2$



- Reaction temperature determined as first deviation of the 1st derivative of weight loss from baseline
- Reaction temperature at log(pO₂)=-2.3 is 541.5°C



 $3\text{LiMn}_2\text{O}_4 \leftrightarrow 3\text{LiMnO}_2 + \text{Mn}_3\text{O}_4 + \text{O}_2$





Thermodynamics of lithium ion batteries Hans J. Seifert

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Figure S1. Calculated Li-Fe-P ternary phase diagram equilibrated with an oxygen potential under reducing conditions. The 'A' arrow indicates off-stoichiometry induced in the samples described in this Letter. The 'B' and 'C' arrow respectively indicate the more common one-to-one Fe/P deficiency and lithium deficiency. The shaded area in phase diagram indicates relevant coating compositions.

Kang and Ceder (2009)

Li-Fe-P-O System



Figure 6. Modified Ellingham diagram for reduction reactions in the Li-Fe-P-O₂ system.

Ong et al. (2008)



Figure 6. Phase diagram of LiFePO₄ (T, for triphylite) and FePO₄ (H, for heterosite) phases showing their merging to a solid solution (D, for disordered) in a eutectoid system. Data points at 25 °C are based on published work by Yamada [10]. The eutectoid point is around the composition x = 0.6 and temperature 200 °C. Above 200 °C, mixtures of heterosite or triphylite and the disordered phase were seen up to around 300 °C. Above 300 °C, the disordered phase dominates.

Li-Fe-P-O System

Dodd et al. (2006)





Detailed phase relationships in the subsystem $LiMn_2O_4 - Li_4Mn_5O_{12} - Li_2Mn_4O_9$ (Yonemura et al. 2004).

p(0₂)=0.21 atm



[1999Pau] Paulsen and Dahn, Chem. Mater., 11, 3065-3079, 1999

Lithium Ion Battery

discharging



material of anode oxidized / material of cathode reduced

 $LiC_6 \rightarrow Li_{1-x}C_6 + xLi^+ + xe^-$

 $Li_{1-x}(Ni, Mn, Co)O_2 + xLi^+ + xe^- \rightarrow Li(Ni, Mn, Co)O_2$

charging



material of anode reduced / material of cathode oxidized

 $Li_{1-x}C_6 + xLi^+ + xe^- \rightarrow LiC_6$

 $Li(Ni,Mn,Co)O_2 \rightarrow Li_{1-x}(Ni,Mn,Co)O_2 + xLi^+ + xe^-$

 $Li_{1-x}C_6 + Li(Ni, Mn, Co)O_2 \leftrightarrow Li_{1-x}(Ni, Mn, Co)O_2 + LiC_6$



R. Huggins, Advanced Batteries



Heat generation rates

$$Q = \Delta G + T\Delta S + W_{el}$$
$$\Delta G = -nFE_{eq} \qquad \Delta S = nF\frac{dE_{eq}}{dT} \qquad W_{el} = -nFE$$

$$q = I\left[\left(E_{eq} - E\right) + T\frac{dE_{eq}}{dT}\right] + q_p$$

Sources of heat generation:

- 1. The "reversible" heat released (or absorbed) by the chemical reaction of the cell
- 2. The "irreversible" heat generation by ohmic resistance and polarisation
- The heat generation by "side reactions",
 i.e. parasitic/corrosion reactions and "chemical shorts"

$$q_{rev} = IT \frac{dE_{eq}}{dT}$$

$$q_{irrev} = I(E_{eq} - E)$$

$$q_p$$

C_{p, eff} measurement on a 40Ah pouch cell



$$c_p = \frac{q}{m\Delta T} \qquad q = \int U \cdot I dt$$



J.R. Selman, S. Hallaj , J. Power Source 97-98, p.726-732 (2001)

Enthalpy of Drop Solution of $Li_{1+x}Mn_{2-x}O_{4-\delta}$



Li-Mn-O Temperature-Composition Ratio Section



• The homogeneity range of the spinel phase determined using thermogravimmetric analysis at $po_2=0.2$ atm

Li-rich boundary of the homogeneity range Li_{1+x}Mn_{2-x}O₄ should be refined





Li-Mn-O System, sample preparation



- Decomposition of LiMn₂O₄ in air
- Samples heat treated at 15 hours and quenched in liquid nitrogen



In-situ technique "entropymetry"



Note: Half cells measured

Yazami et al. in Lithium Ion Rechargeable Batteries, WILEY-VCH (2010)

$\text{Li}_{x}\text{Mn}_{3-x}\text{O}_{4}$ Spinel Li, Mno, LiMnO₂ Mn_0 LiMn₂04 Temperature Li₂MnO₃ в 1000°C- T_2 cubic Spinel Spinel ≈960°C letrogonal Spinel 11x+ 0 1 $T_{c1}(r_1)$ 880°C С 04+13 800°Calston NN 2*+5 II: Spinel + Li2MnO3 3.5444 1 Т A M10 2 3 Mn₂0 T_{c1}(r) III: Spinel + Mn₂O₃ I: Spinel 600°C-T_{cL}(r1) R-MAO 2 Spinel 03 Mes $T_{cL}(r)$ D T_a ≈430°C LiMn₂O₃ Mn₂O₃ MnO₂ Mn₂O₃ 400°C-Li_{0.33}MnO₂ 0.6 0.0 0.4 0.5 0.1 Li₄Mn₅O₁₂ 0 T (°C) r₁ 2 r3 r_2 Li₂MnO₃ LiMn₂04 \rightarrow $r = n_{Li}/n_{Mn}$ Li/(Li+Mn) Luo and Martin, 2007 λ -MnO₂ LiMnO₂ LiMn₂O₄ Paulsen and Dahn, 1999 Li1-8"Mn2O4-8 Li₄Mn₅O₁₂ LiMn₂O₄ Li-Mn-O System Li2MnO3 Lattice paramter Lattice paramter decreases decreases Charge-discharge capacity decreases Cycling property increases $Li_{1-\delta''}Mn_{2-\delta'}O_4$ Li2Mn4O9 $Li_{1+x}Mn_{2-x}O_4$ Yonemura et al. 2004 Li₄Mn₅O₁₂

2nd kind phase diagram in the Li-Mn-O system
German Research Foundation, Priority Program 1473, Materials with New Design for Improved Lithium Ion Batteries -WeNDeLIB

Battery Properties	Thermodynamics and Kinetics
Thermal runaway	Oxygen partial pressure, Gibbs free energies of reactions
Voltage, potential	Chemical potentials (of lithium)
Capacity, energy- and power density	Phase diagrams, Gibbs free energies
Life time	Stability of compounds in battery; Materials constitution
Power- and materials loss during first charge cycle	Formation of SEI; Relative thermochemical stabilities of materials for electrodes and electrolyte

Relationships Thermodynamics and Electrochemistry

Total change in enthalpy and entropy between two electrode compositions x_1 and x_2 :

$$\overline{\Delta S}\Big|_{x1}^{x2} = F\left(\int_{x1}^{x2} \frac{\partial E_0(x,T)}{\partial T}\Big|_x\right)$$
$$\overline{\Delta H}\Big|_{x1}^{x2} = F\int_{x1}^{x2} \left(-E_0(x,T) + T\frac{\partial E_0(x,T)}{\partial T}\Big|_x dx\right)$$

... and with normalization

$$y = \frac{x}{\Delta x_{\text{max}}}, \ 0 < y < 1$$

$$\overline{\Delta S}\Big|_{0}^{1} = F\left(\int_{0}^{1} \frac{\partial E_{0}(y,T)}{\partial T}\Big|_{y} dy\right)$$

$$\overline{\Delta H}\Big|_{0}^{1} = F \int_{0}^{1} \left(-E_{0}(y,T) + T \frac{\partial E_{0}(y,T)}{\partial T} \Big|_{y} dy \right)$$

Thermodynamic functions of active materials are needed



(2) FreedomCAR and Fuel Partnership and United States Advanced Battery Consortium (USABC), Electrochemical Energy Storage Technical Team Technology Development Roadmap (Southfield, MI: USCAR, 2006)



M Winter, J. O. Besenhard, Chem. uns. Zeit 33, p. 320-332 (1999)

Li-Cu-Fe-O System



- Thermodynamic calculations based on the CALPHAD method
 - predict battery performance

equilibrium voltages (OCV) plateau capacities

- Database development for the Li-Cu-Fe-O System
 - The Cu-Fe-O ternary system assessed by Khvan et al.
 - First calculated phase diagrams in the Li-Cu-O system addressed in present work

N. Saunders, I. Ansara (Ed), Cost 507, Report,1994,168–169

B. Hallstedt, L.J. Gauckler CALPHAD, 2003, 27:177-191

K. Chang, B. Hallstedt, CALPHAD, 2011, 35:160-164



Isothermal calorimetric measurements on a 16 mAh Lithium ion coin cell

Karlsruhe Institute of Technology

Discharge parameters:

- Method: constant current (CC)
- $U_{min} = 2.75 V$
- I = 16 mA \rightarrow 1C-rate

Charge parameters:

- Method: constant current, constant voltage

(CCCV)

- $U_{max} = 4.25 V$
- I = 16 mA \rightarrow 1C-rate
- $I_{min} = 1.6 \text{ mA}$



Isothermal Battery Calorimeter (IBC)

Cell type: coin cell LIR 2016, Conrad energy (commercial) Capacity: 20 ± 5 mAh; Working voltage: 3.6 V



Isothermal calorimetric measurements on a 16 mAh Lithium ion coin cell





Accelerating Rate Calorimeter (ARC)

ARC provides an adiabatic environment in which a sample may be studied under conditions of negligible heat loss

heat of reaction: $\Delta H = \Delta T_{ad} c_p$

total heat generated: $Q_{\Sigma} = m\Delta H$





ESARC: Ø: 10cm h: 10cm

Thermodynamics of electrochemical reactions





$$\Delta G = -nFE$$

Gibbs-Helmholtz equation

$$\Delta S = -\left(\frac{\delta \Delta G}{\delta T}\right)$$

entropic change of electrochem. reaction

$$\Delta S = nF\left(\frac{\delta E}{\delta T}\right)$$

reversible heat

$$Q_{rev} = T\Delta S = nFT\left(\frac{\delta E}{\delta T}\right)$$

S. Tobishima, J.Yamaki, J. Power Source 81-82, p. 882–886 (1999)

A.K. Shukla, T.P. Kumar, Current Sci. 94, p. 314-331 (2008)

Enthalpy of Drop Solution of $Li_{1+x}Mn_{2-x}O_{4-\delta}$

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Colver Colorimeter			
		•	
		/	

Al2O3 plug	• • • • •
SiO ₂ glass dropping tube for sample introduction	
Platinum tube for bubbling gas	
Platinum crudible where solvent is introduced	
SiO2 glass crucible	
SiO2 glass liner	•
Inconel protection tube	Y
Sodium Molybdate, 7	/ 00°C

AlexSys 1000, SETARAM High Temperature Calvet Calorimeter

Enthalpy of Drop Solution of $Li_{1+x}Mn_{2-x}O_{4-\delta}$



Sample Number	DROP 1	DROP 2	DROP 3	DROP 4	DROP 5	DROP 6	DROP 7	DROP 8
Date	08. Mai	09. Mai	09. Mai	09. Mai				
Mass pellet (mg)	6,00	5,14	6,10	6,62	4,85	5,35	4,83	5,59
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T(cal.) (°C)	700,40	700,40	700,40	700,40	700,40	700,40	700,40	700,40
Formula weight (g/mol)	180,815	180,815	180,815	180,815	180,815	180,815	180,815	180,815
Moles of LiMn2O4 (mol)	0,0000332	0,0000284	0,0000337	0,0000366	0,0000268	0,0000296	0,0000267	0,0000309
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Measured Heat Effect (kJ/mol)	255,1652	250,0926	241,8308	241,0849	241,5781	241,2203	241,6957	240,8258
Accepted Measurement	0	0	1	1	1	1	1	1

$\text{Li}_{x}\text{Mn}_{3-x}\text{O}_{4}$ Spinel Li, Mno, LiMnO₂ Mn_0 LiMn₂04 Temperature Li₂MnO₃ в 1000°C- T_2 cubic Spinel Spinel ≈960°C letrogonal Spinel 11x+ 0 1 $T_{c1}(r_1)$ 880°C С 04+13 800°Calston NN 2*+5 II: Spinel + Li2MnO3 3.5444 1 Т A M10 2 3 Mn₂0 T_{c1}(r) III: Spinel + Mn₂O₃ I: Spinel 600°C-T_{cL}(r1) R-MAO 2 Spinel 03 Mes $T_{cL}(r)$ D T_a ≈430°C LiMn₂O₃ Mn₂O₃ MnO₂ Mn₂O₃ 400°C-Li_{0.33}MnO₂ 0.6 0.0 0.4 0.5 0.1 Li₄Mn₅O₁₂ 0 T (°C) r₁ 2 r3 r_2 Li₂MnO₃ LiMn₂04 \rightarrow $r = n_{Li}/n_{Mn}$ Li/(Li+Mn) Luo and Martin, 2007 λ -MnO₂ LiMnO₂ LiMn₂O₄ Paulsen and Dahn, 1999 Li1-8"Mn2O4-8 Li₄Mn₅O₁₂ LiMn₂O₄ Li-Mn-O System Li2MnO3 Lattice paramter Lattice paramter decreases decreases Charge-discharge capacity decreases Cycling property increases $Li_{1-\delta''}Mn_{2-\delta'}O_4$ Li2Mn4O9 $Li_{1+x}Mn_{2-x}O_4$ Yonemura et al. 2004 Li₄Mn₅O₁₂

2nd kind phase diagram in the Li-Mn-O system



Chemical potential diagram in the Li-Mn-O system (Tsuji et al. 2005).

What to do next: (1) Evaluation; (2) Solution phase modeling; (3) Thermodynamic optimization





- LiMn₂O₄ composition is a vertical line in the temperature-composition ratio diagram
- p(O₂)=0.21 atm is a horizontal line in the potential diagram



 $p(O_2)=0.21 \text{ atm}$ Li/Mn ratio for LiMn₂O₄



 $LiMn_2O_4 \leftrightarrow zLi_2MnO_3 + Li_{1-2z}Mn_{2-z}O_{4-3z-y}(tet) + (y/2)O_2$

 $\text{Li}_{1-2z}\text{Mn}_{2-z}\text{O}_{4-3z-y} \text{ (tet) } \leftrightarrow \text{Li}\text{Mn}\text{O}_2 + (1/3) \text{Mn}_3\text{O}_4 + \{(1/3)-(y/2)\}\text{O}_2$

[1999 Pau] Chem. Mater., 11 (1999), 3065-3079. [2005 Tsu] J. Phys. Chem., Solids. 66 (2005), 283-287.

Commercial cathode material LiMn₂O₄

Spinel structure



Gravimetrical energy density (capacity)



Figure 6. Phase diagram of LiFePO₄ (T, for triphylite) and FePO₄ (H, for heterosite) phases showing their merging to a solid solution (D, for disordered) in a eutectoid system. Data points at 25 °C are based on published work by Yamada [10]. The eutectoid point is around the composition x = 0.6 and temperature 200 °C. Above 200 °C, mixtures of heterosite or triphylite and the disordered phase were seen up to around 300 °C. Above 300 °C, the disordered phase dominates.

Dodd et al. (2006)

Li-Fe-P-O System

Lithium Ion Batteries – Operation





Figure 6. Modified Ellingham diagram for reduction reactions in the Li-Fe-P-O₂ system.

Ong et al. (2008)



(b) μ_{O1} = -11.52 eV, LiFePO₄ appears

Figure 2. Phase diagrams at less reducing environments.



Accidents with lithium ion batteries



Rechargable battery of a laptop – overheating or internal short circuit



stationary energy storage – overcharging



Fire 3 days after crashtest with EV – mechanical impact