

ab initio Computational Materials Science

Ralf Stoffel and Richard Dronskowski

Introduction to Quantum Mechanics: the Wave Function

Total Energy Calculations

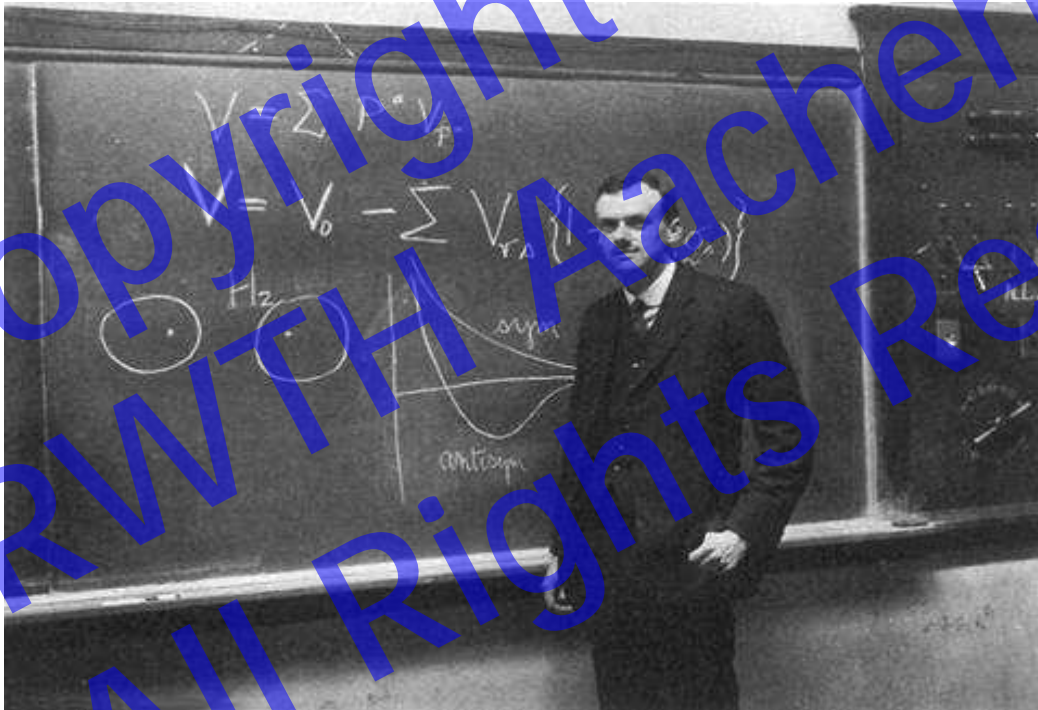
Application to “Real-World” Problems

Temperature

Summary & Outlook

Where everything starts

"The underlying physical laws necessary for the mathematical theory of a large part of physics and the whole of chemistry are thus completely known, and the difficulty is only that the exact application of these laws leads to equations much too complicated to be soluble."



**Paul Adrien Maurice Dirac
(1929)**

The Wave Function and Schrödinger's Equation

$$\hat{H} \Psi = E \Psi$$

$\Psi = \Psi(\mathbf{R}_i, \mathbf{r}_i, t)$ is the entire atomic-like system
the wave function Ψ contains the **information in its entirety**, such as

- chemical constitution & configuration
- structures
- energies
- forces
- chemical bonding

and may be computed „from first principles“ (*ab initio*)

Pioneers of Physics and Chemistry



Heitler



London



Bloch



Mulliken



Hoffmann

2008



Parrinello



Schrödinger



Slater



Pople



Hellmann



Pauling



Kohn

1926

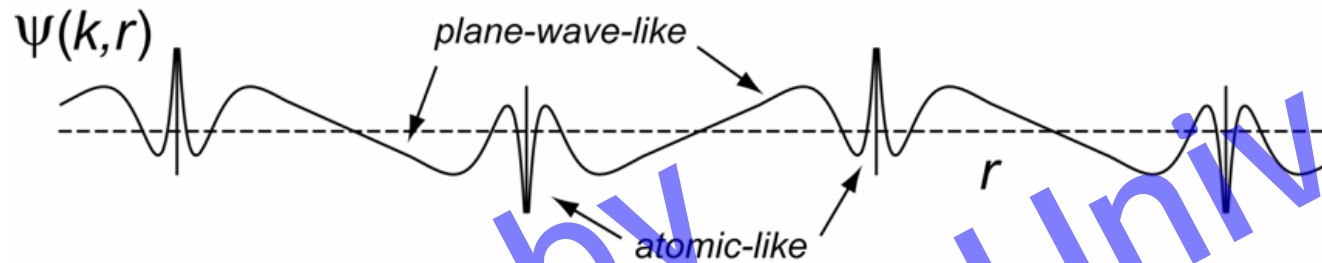


Dirac



Fock

Reasons for Computational Materials Science



a reliable wave function (and its energy) gives access to

structural properties (lattice & spatial parameters)

electronic properties (band structures, densities-of-states, chemical bonding analysis, magnetism...)

thermodynamic properties (total energies, heats of formation, phonon dispersion → **free energy, entropy, heat capacity**)

Also: dynamic, optical, mechanical and other properties

Theoretical Calculation of Formation Enthalpy

$$\Delta G = \Delta H - T\Delta S \quad \begin{array}{l} \text{solid as} \\ \approx \\ \text{a rock} \end{array} \quad \Delta H \approx \Delta E$$

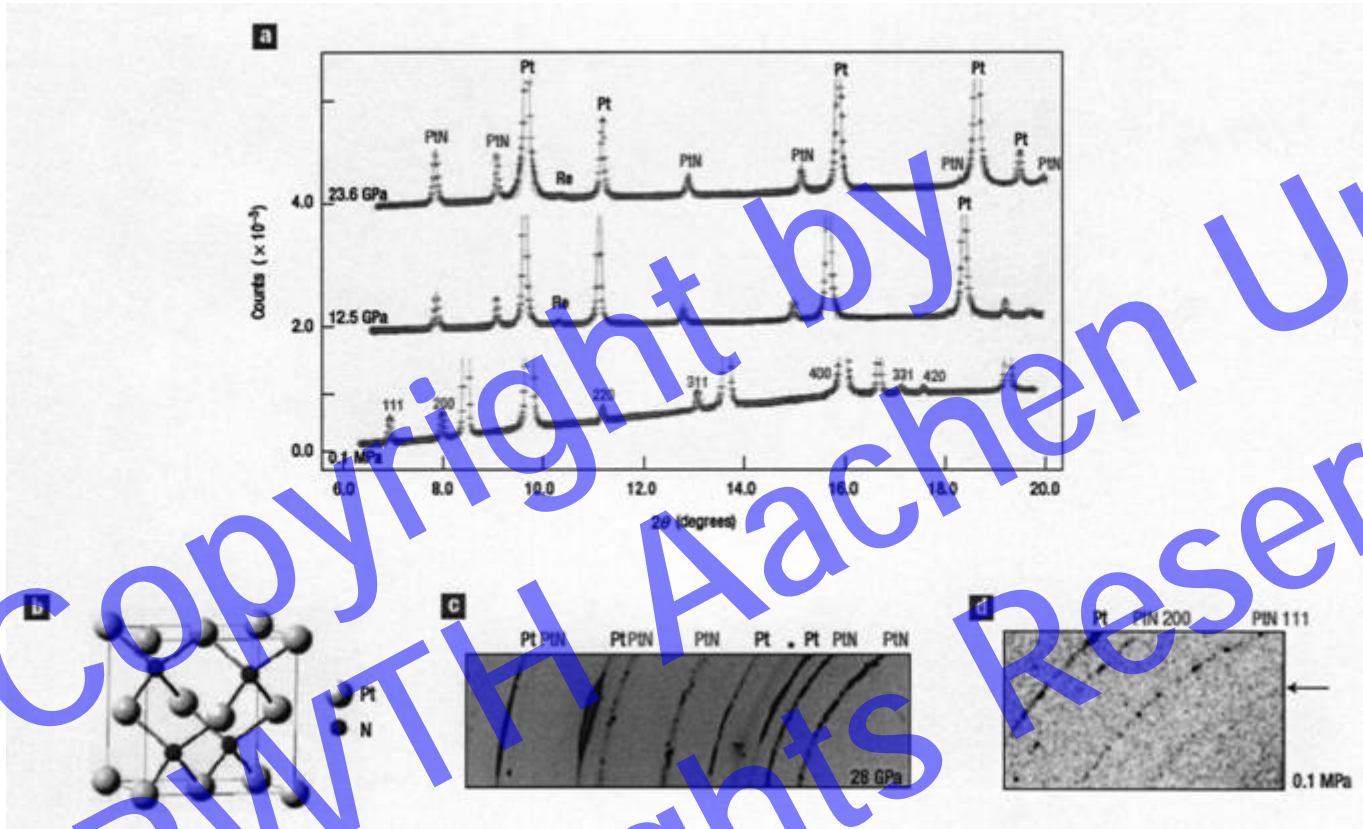
density-functional (first-principles) total-energy calculations

E - V diagrams $\rightarrow p = -\partial E/\partial V \rightarrow H = E + pV$

finally $\Delta H(p)$

- \rightarrow stability of solid compounds / different structure types
- \rightarrow prediction of high-pressure phases

Platinum Nitride: "PtN"



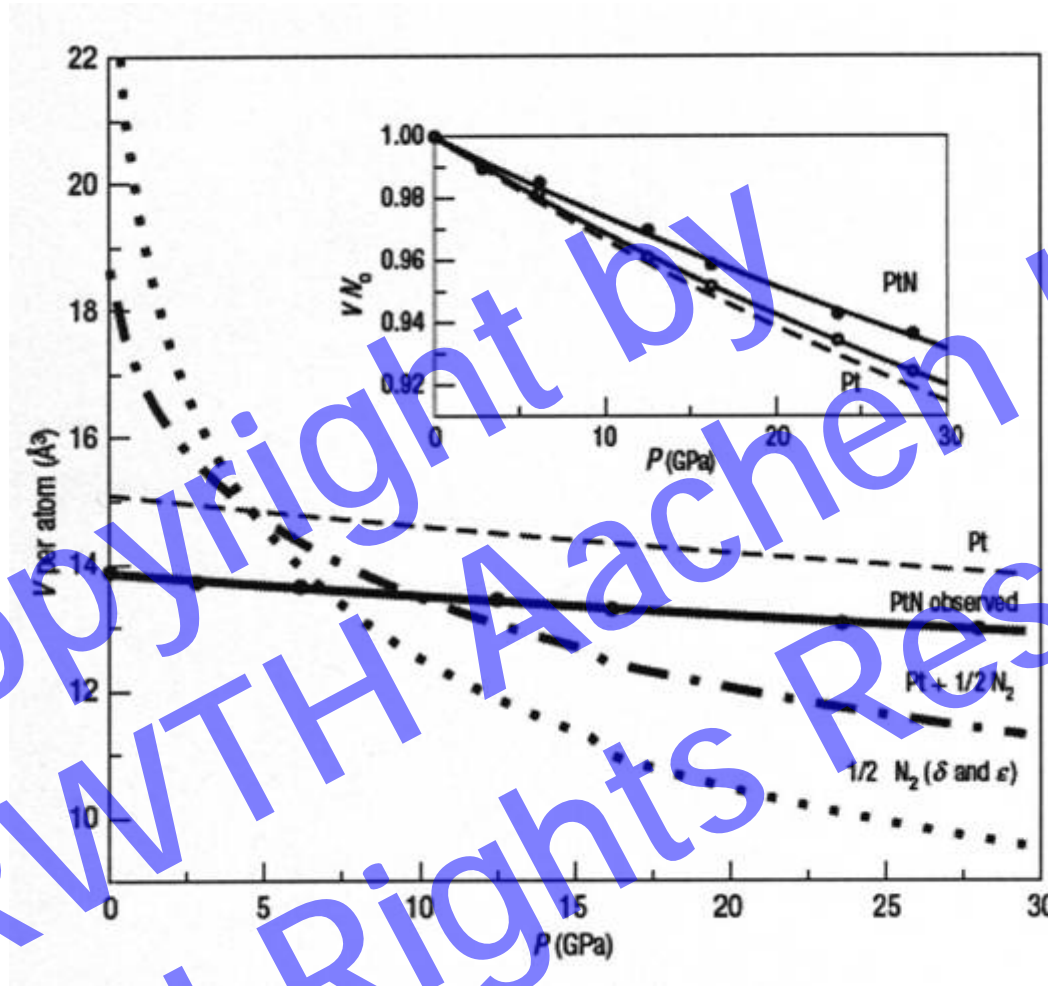
from Pt + N₂
at $p > 45$ GPa
and $T > 2000$ K

X-ray data:
fcc unit cell with
 $a = 4.8041$ Å,
[ZnS] type,
 $B_0 = 372 \pm 5$ GPa

chemical analysis:
 PtN_{1-x} with $x < 0.05$

E. Gregoryanz, C. Sanloup, M. Somayazulu,
J. Badro, G. Fiquet, H.-K. Mao, R. J. Hemley,
Nature Mater. 2004, 3, 294

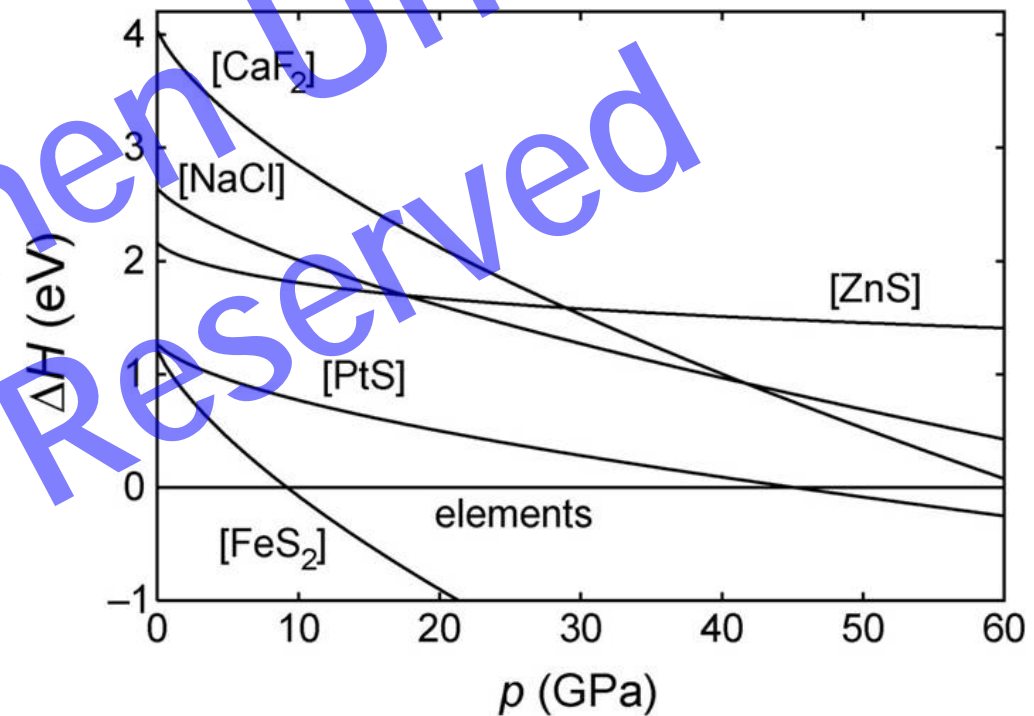
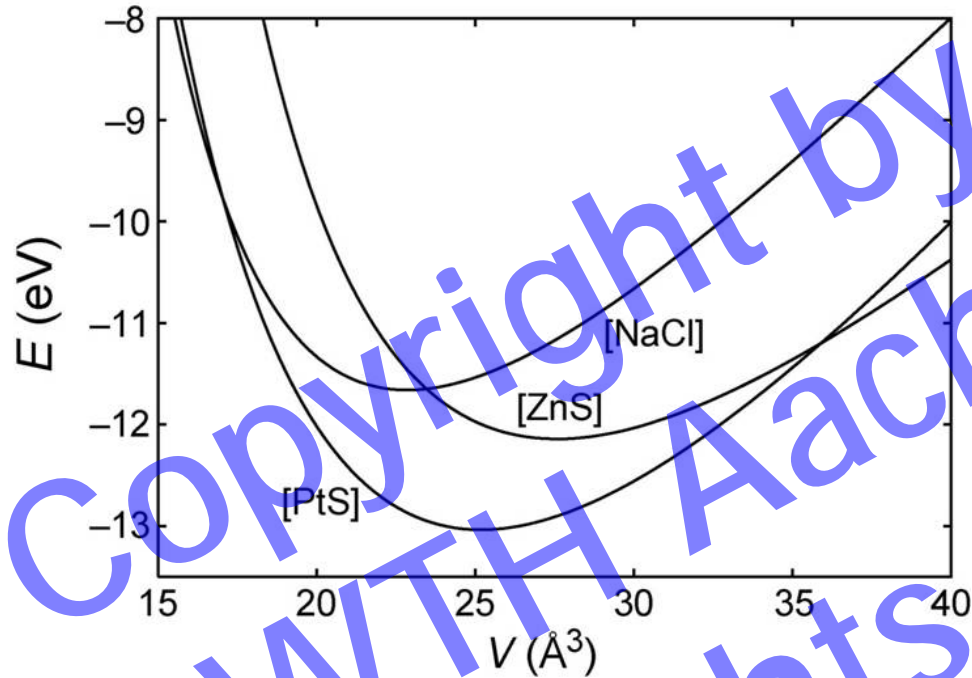
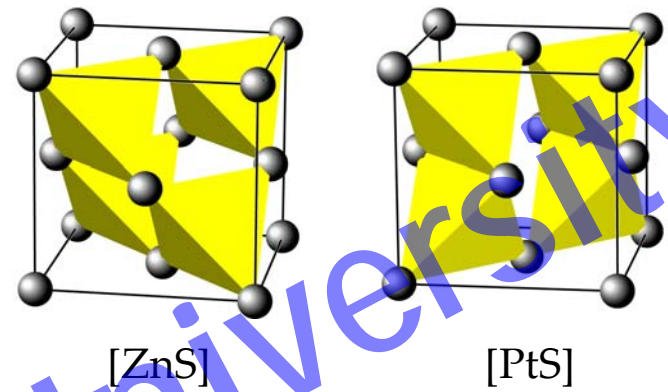
Platinum Nitride: the Obvious Discrepancy



product phase is less dense than the elements!

Le Chatelier's principle violated

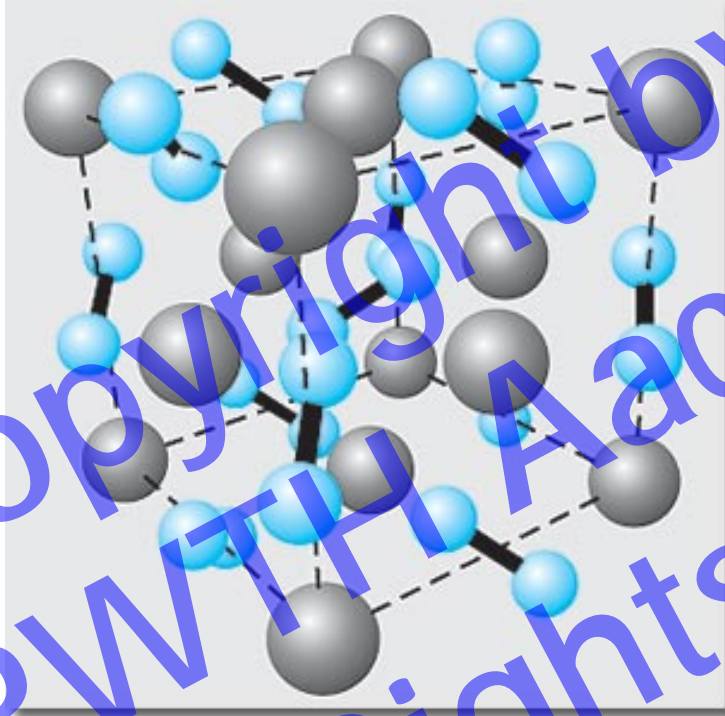
Platinum Nitride: E (V), H (p)



J. von Appen, R. Dronskowski,
Angew. Chem. Int. Ed. **2006**, *45*, 4365

“Platinum Nitride”: Experimental Realization

by repeating the 2004 experiments of Gregoryanz et al.:



composition: PtN₂, *not* PtN
pyrite [FeS₂] structure type
lattice parameter $a \approx 4.8 \text{ \AA}$
very large bulk modulus

...but they still call it a “nitride”

J. C. Crowhurst, A. F. Goncharov, B. Sadigh, C. L. Evans,
P. G. Morrall, J. L. Ferreira, A. J. Nelson
Science **2006**, 311, 1275

ab initio Thermochemistry

Thermodynamic properties including **temperature**:
lattice vibrations (phonons) → **Gibbs free energy**

$$G(T, p) = F(T, V) + pV$$

$$F(T, V) = E(V) + F_{ph}(T, V)$$

$F_{ph}(T, V)$: *phonon free energy*

$$F_{ph} = \frac{1}{2} \sum_{\vec{k}, \nu} \hbar \omega(\vec{k}, \nu) + k_B T \sum_{\vec{k}, \nu} \ln \left[1 - \exp \left(\frac{-\hbar \omega(\vec{k}, \nu)}{k_B T} \right) \right]$$

we only need to know the *phonon frequencies* ω

⇒ also **heat capacity** and **entropy**

How to obtain the phonon frequencies

1. Calculation of the **forces** F_α acting on the atoms when the other atoms are slightly ($u_\beta \approx 0.01 \text{ \AA}$) dislocated

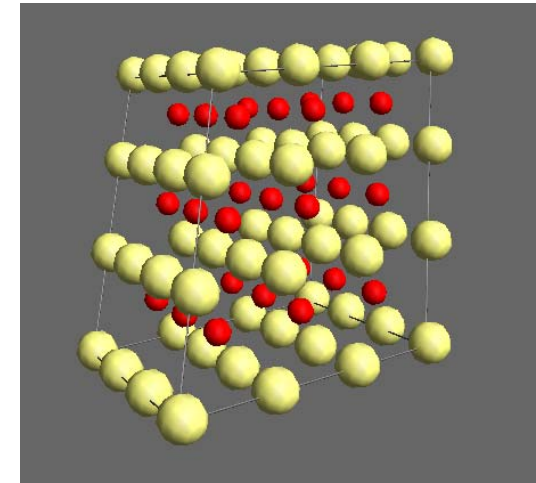
2. **force constant matrix** $\Phi_{\alpha\beta}$ $\Phi_{\alpha\beta} = -\frac{F_\alpha}{u_\beta}$

3. **dynamical matrix** $D_{\alpha\beta}$ $D_{\alpha\beta}(jj', \vec{k}) = \frac{1}{\sqrt{m_j m_{j'}}} \sum \Phi_{\alpha\beta} \exp[i\vec{k}(\vec{r} - \vec{r}')]]$

4. **diagonalisation** of $D_{\alpha\beta} \rightarrow \omega$

$$|D_{\alpha\beta}(jj') - \delta_{\alpha\beta} \delta_{jj'} \omega^2| = 0$$

problem: supercells (> 50 atoms),
high demand of computational time
and memory (scaling $\approx N^{2-3}$)



Project ELSA

Together with  and others:

Computational modelling and preparation of the

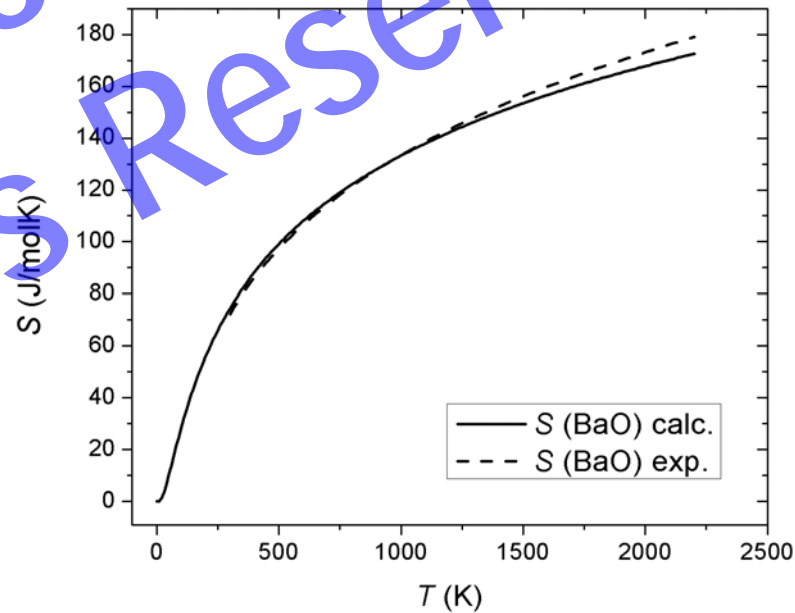
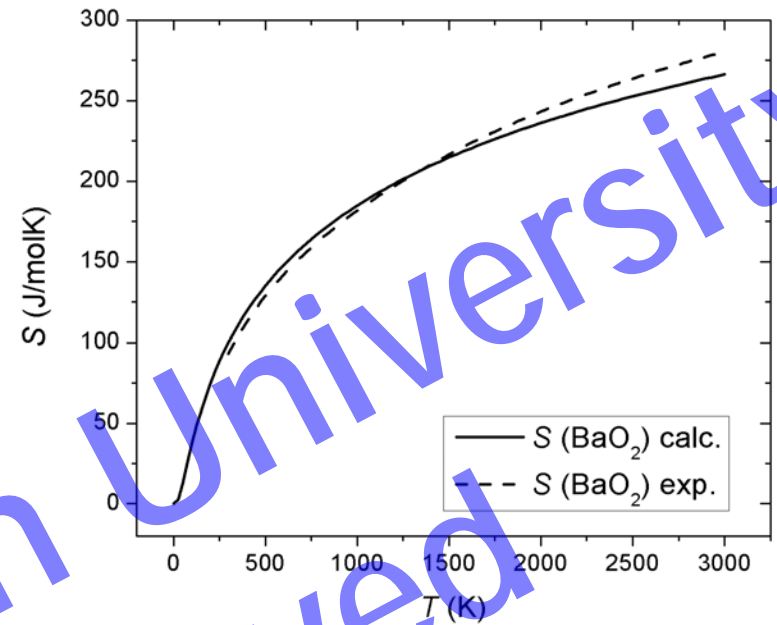
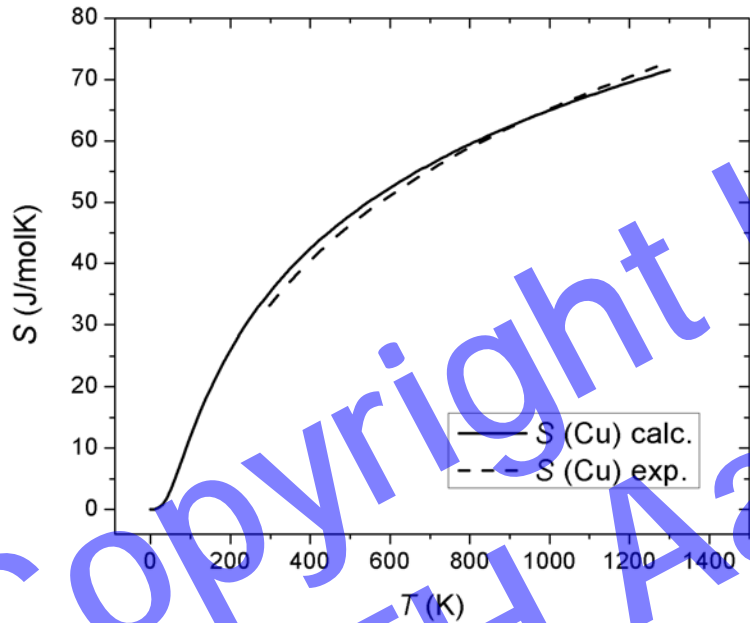
high temperature superconductor $\text{YBa}_2\text{Cu}_3\text{O}_{6+x}$

Our (quantum-chemical) part:

Prediction of thermochemical potentials where there is no experimental data available

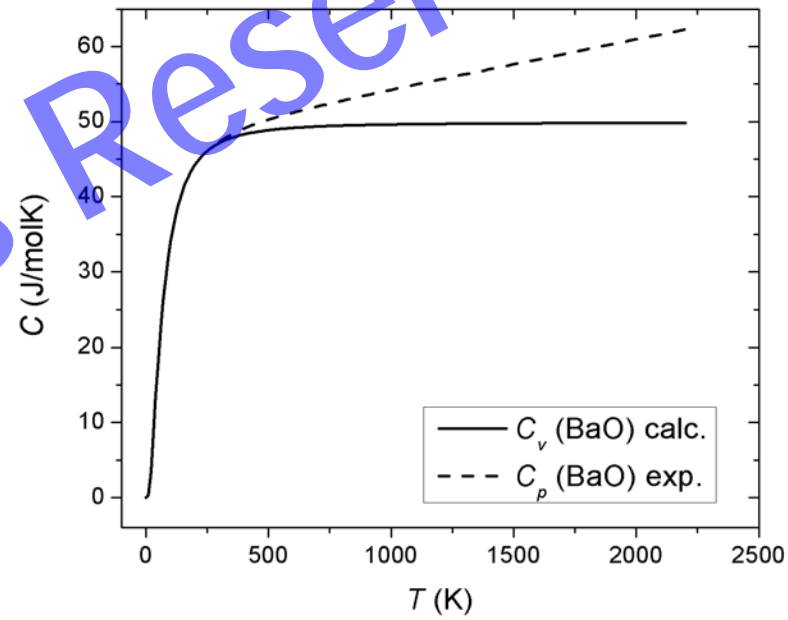
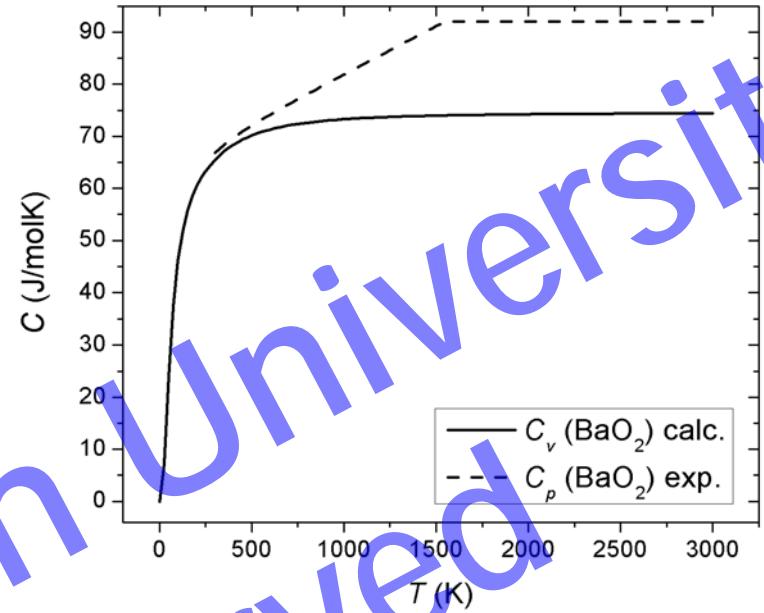
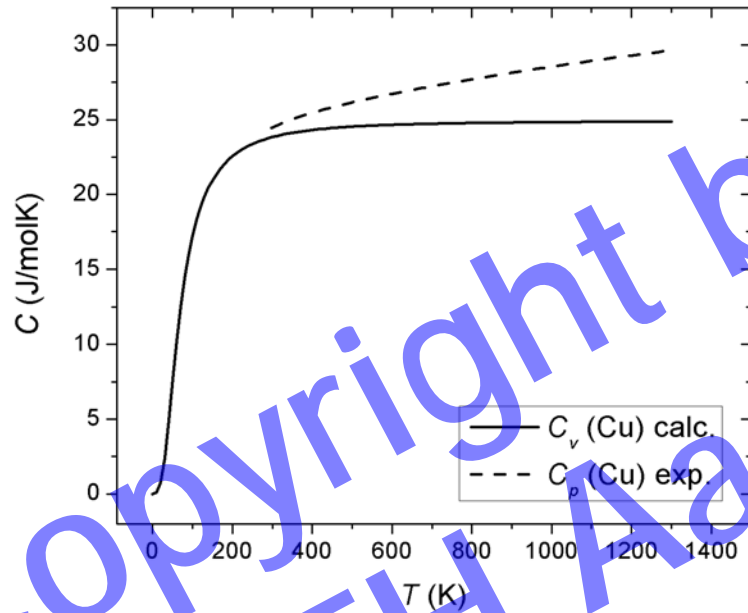
First step: comparing properties of compounds with known thermochemical properties (e.g. Cu, CuO, BaO, BaO₂...)

Entropy



Good reproduction of experimental data, only small errors

Heat Capacity



Calculation of C_v not C_p

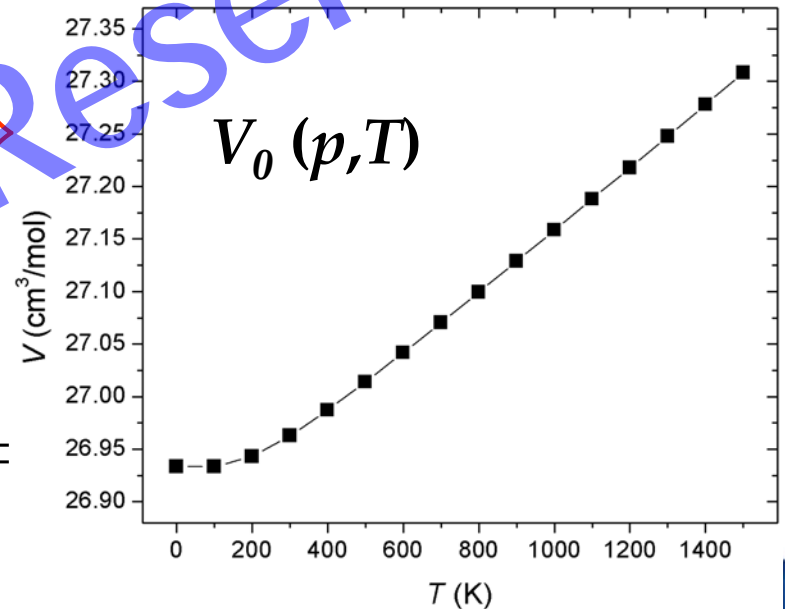
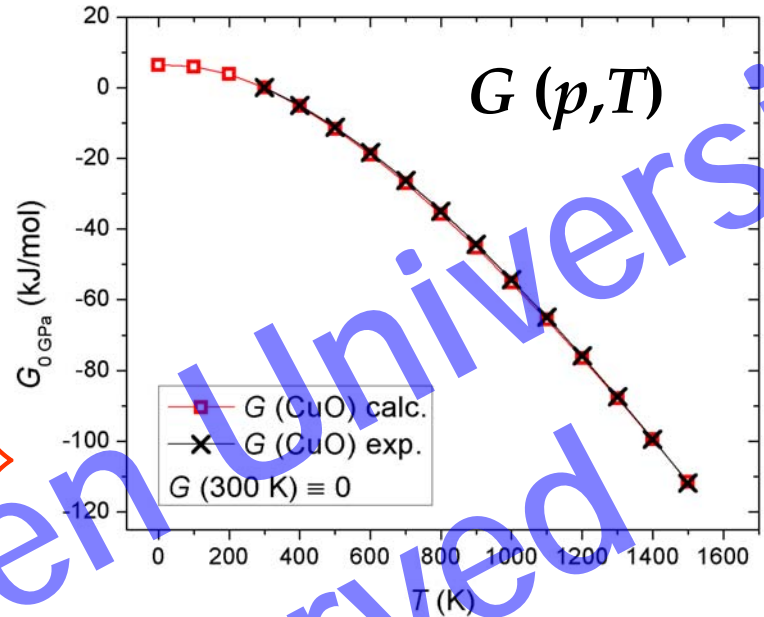
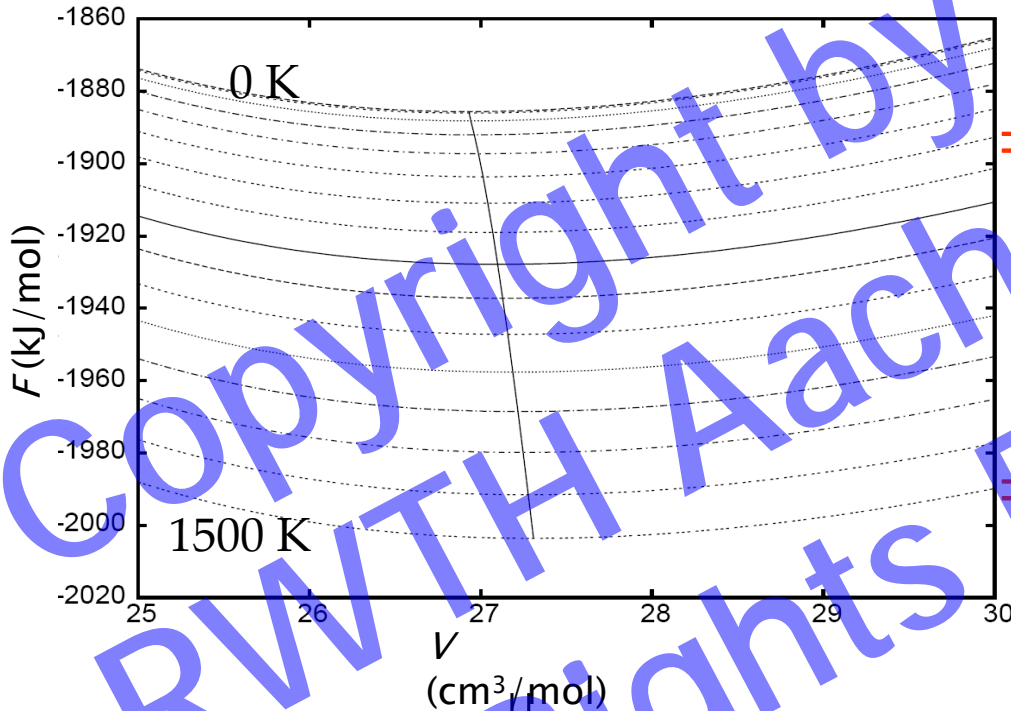
⇒ correction:

$$C_p - C_v = \alpha^2 B_T V T \approx \text{const.} \cdot T$$

Free Energy

$F(T)$ at different V (96-110%)

$\Rightarrow F(V)$ at different T

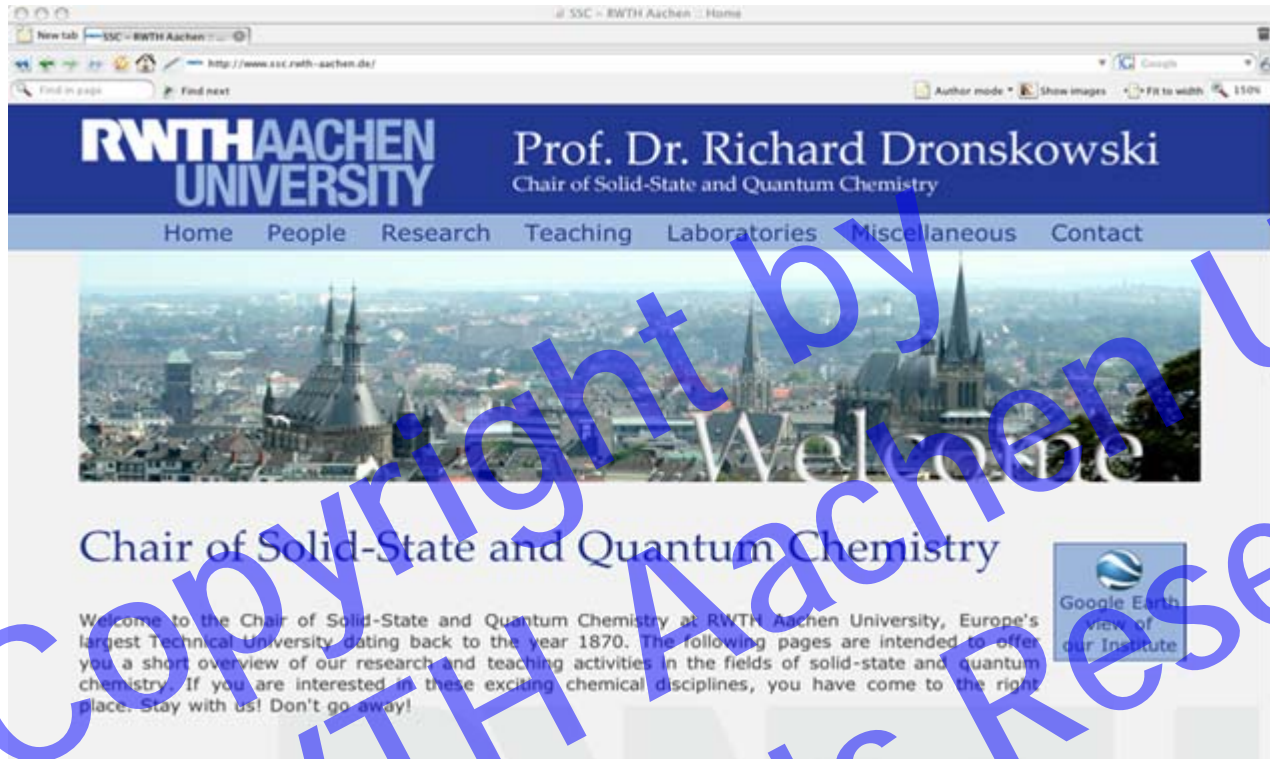


$$C_p = C_V + \alpha^2 B_T VT$$

Summary

- *ab initio* total energy calculations: access to $\Delta H(p)$ at 0 K
- lattice vibrations: *T-dependent* thermochemical properties, access to $G(p,T)$, S , C
- good estimation of *entropy* and *heat capacity* of compounds with known thermodynamical properties
- next steps: properties of $\text{YBa}_2\text{Cu}_3\text{O}_{6+x}$
 - handling of gaseous compounds (e.g. oxygen)?
 - comparing of $\Delta G(p,T)$ with exp. data
 - generate thermochemical data

For more information:



www.ssc.rwth-aachen.de

